

Bioactive Bone Cements – Advantages and Limitations

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UNIVERSITY
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HEA

Higher Education Authority
An tÚdarás um Ard-Oideachas



MSSI

Ceramics for Biomedical Applications



Alumina on Alumina Hip Bearing



Various ceramic implants

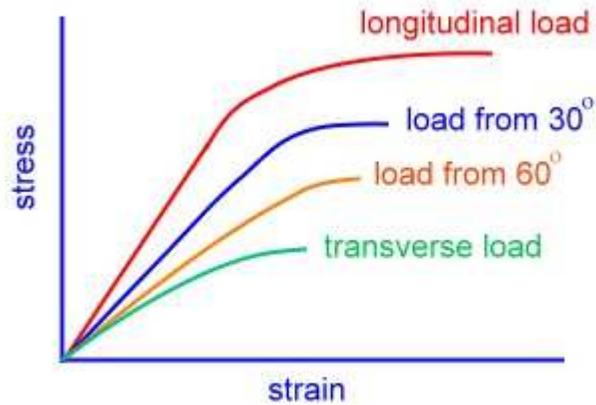


Dental Restoratives

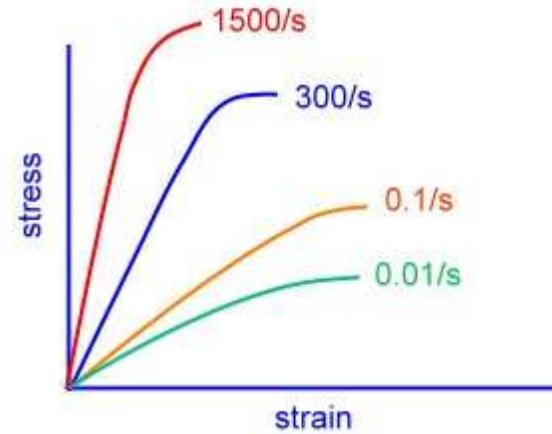


Calcium Phosphate Bone Substitute

Bone Mechanics



Anisotropic Behavior of the Bone



Rate Dependent Behavior

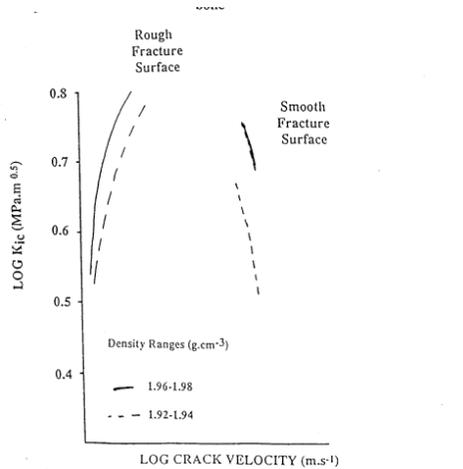


Figure 1d Plot of log stress intensity factor (K_{Ic}) against log crack velocity (V)

V-K Curve Bone

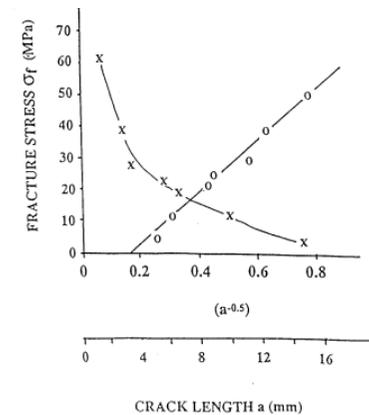
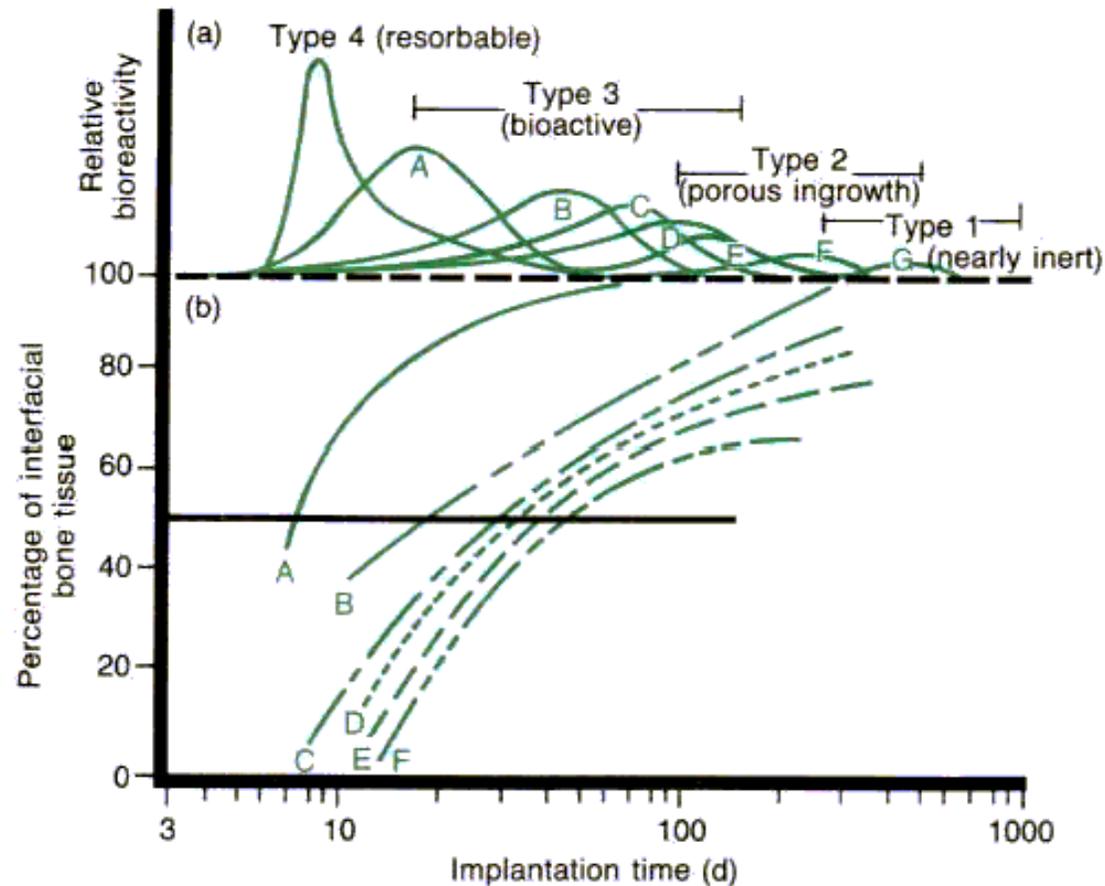


Figure 1e Plot of fracture stress against crack length (a) for samples of cortical bone

σ_f Vs Flaw Size - Bone

Ceramics of Implant Use



Bioactivity spectrum for various bioceramic implants, (A) Relative rate of bioreactivity, (B) Time dependence of formation of bone bonding at an implant interface. ¹

1. Hench, L J. Am. Ceram. Soc. 74 (7) 1487-510 (1991)

Bioactive Material

- “one which has been designed to induce specific biological activity”²
- “Bone bonding..... via a time-dependent, kinetic modification of the surface that occurs upon implantation the surface forms a biologically active hydroxycarbonate apatite (HCA) layer which provides the bonding interface with tissues¹
- “.the essential requirement for a material to bond to living bone is the formation of bone-like apatite on its surface when implantedthis in vivo apatite formation can be reproduced in a simulated body fluid (SBF) with ion concentrations nearly equal to those of human blood plasma.”³

1. Hench, L J. Am. Ceram. Soc. 74 (7) 1487-510 (1991)

2. ESB consensus conference of 1987.

3. Kokubo T, Takadama H. How useful is SBF in predicting in vivo bone bioactivity? Biomaterials 2006;27:2907–15.

“Can bioactivity be tested in vitro with SBF solution?”⁴

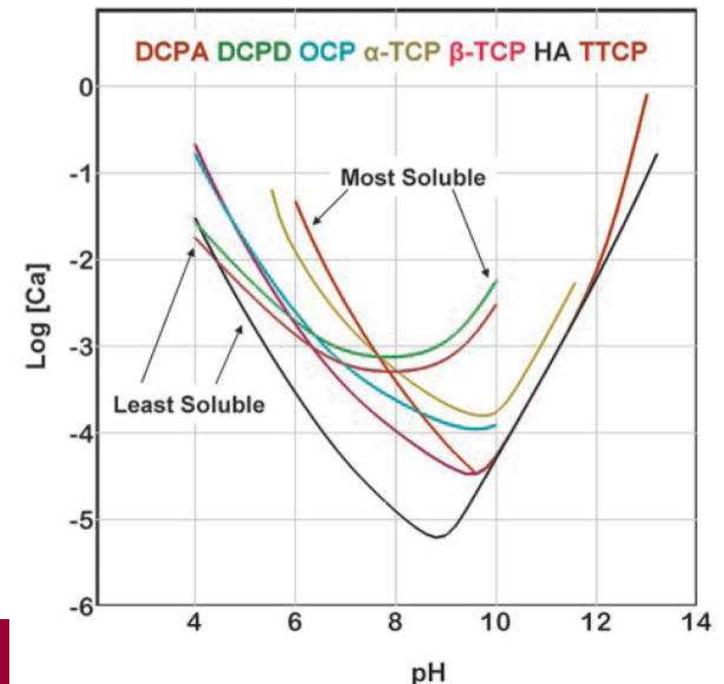
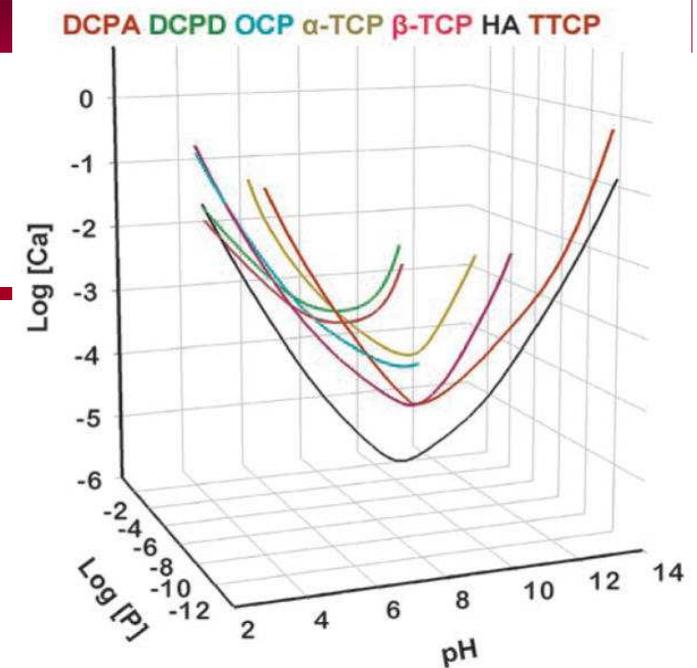
- “Both serum and SBF are supersaturated towards apatite crystals..... system is metastable and will eventually become thermodynamically stable by forming apatite crystals”
- “Bioactivity testing with SBF may lead not only to false positive but also to false negative results”
- “The use of an in vitro protocol for testing the bone bonding potential of a material remains a very attractive concept and should be contemplated very carefully”

Ca P Salts Solubility

- For CaPs this can change with pH, conc. and temp.
- A salt whose isotherm lies below that of another is less soluble

At pH 7 the *in vivo* degradation rate of CaP can be predicted

MCPM > TTCP \approx α -TCP > DCPD > DCPA > OCP > β -TCP > HA



Bioactive Bone Cements

Bone “Cement” differentiated from Bone “Substitute”

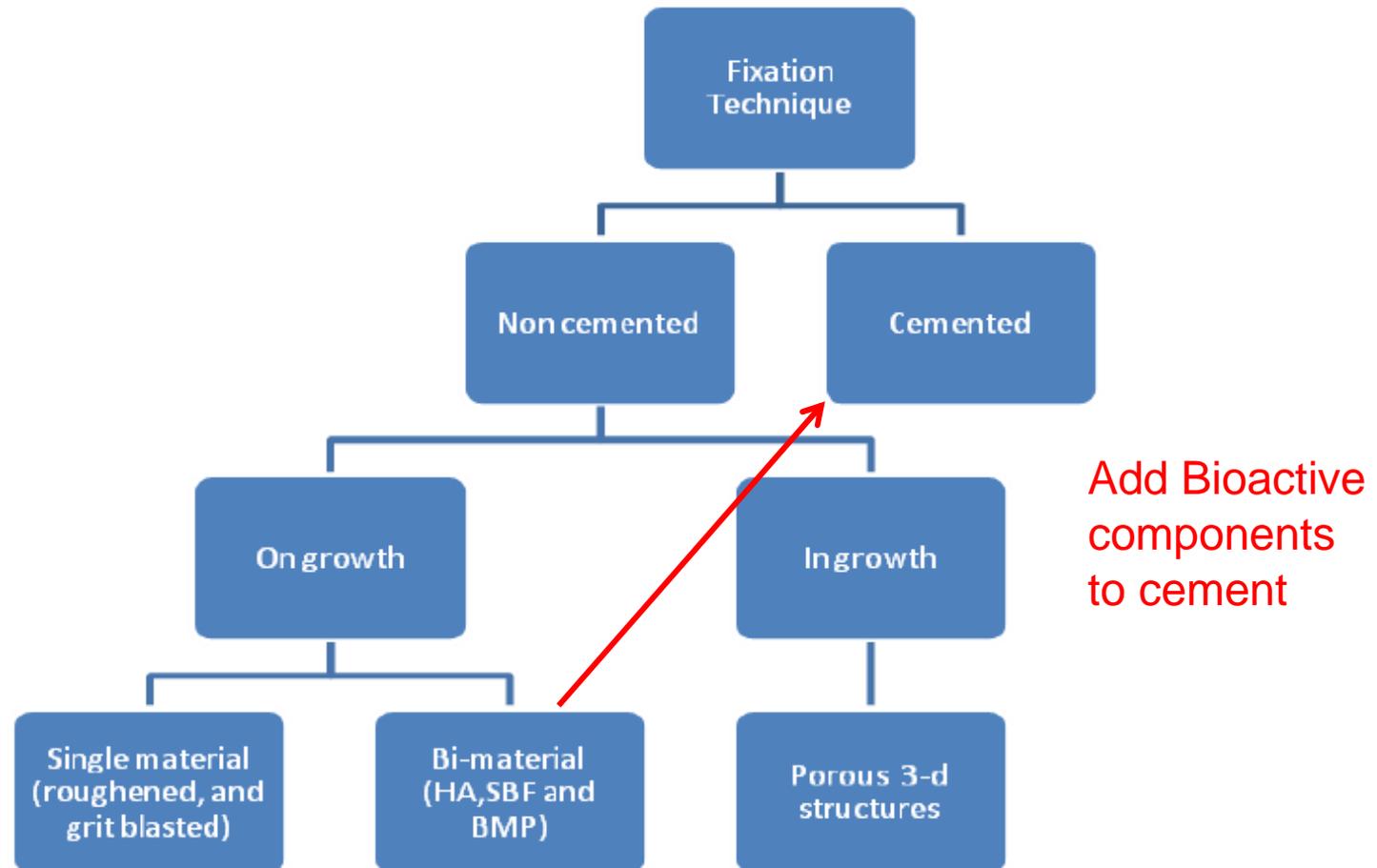
Implies:

- In situ setting – takes up form of individual defect
- Load bearing
- Adhesive (Necessary for stress transmission)
- Bioactive
- Fixation of other devices (Metal/Ceramic/Polymer)

Mutually Exclusive for high strength (density) ceramics

- **FDA Labelling : ”Bone Void Filler intended only for orthopedic applications as a filler for gaps and voids that are not intrinsic to the stability of the bony structure”**

Fixation of Orthopaedic Devices



Design Criteria for Orthopaedic Bone Cement

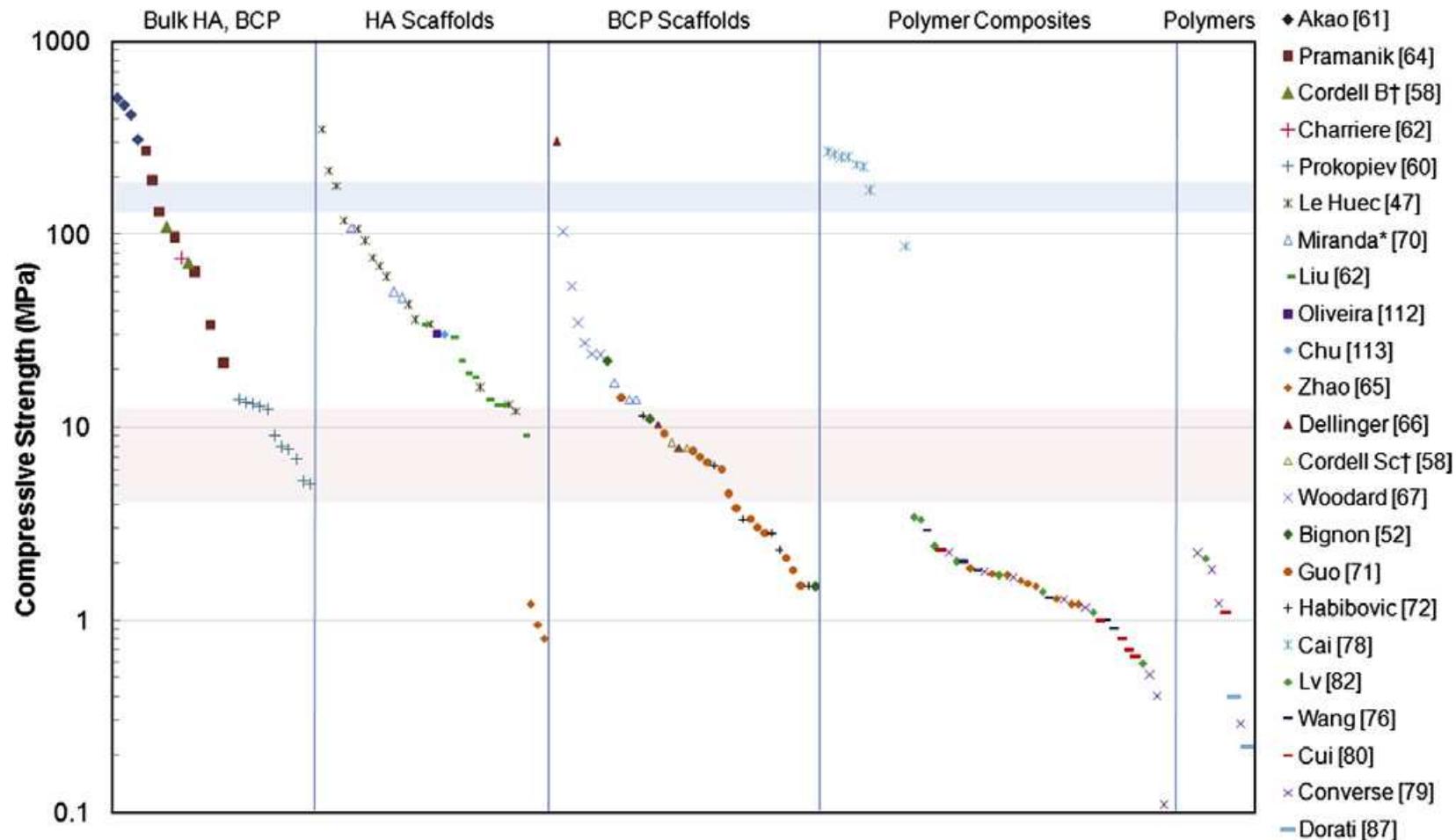
- Ease of Placement and Handling
- Chemical Adhesion - Hydrophilic
- Modulus Match with Bone
- Non-cytotoxic (no foreign body response)
- Bioactive (Osteoconductive / Osteoinductive)
- If resorption required – match regeneration rate
- Rapid Setting
- Dimensionally Stable
- Radiopaque
- Suitable Matrix for drug delivery
- Mechanical & Fracture Properties similar to bone at target

site

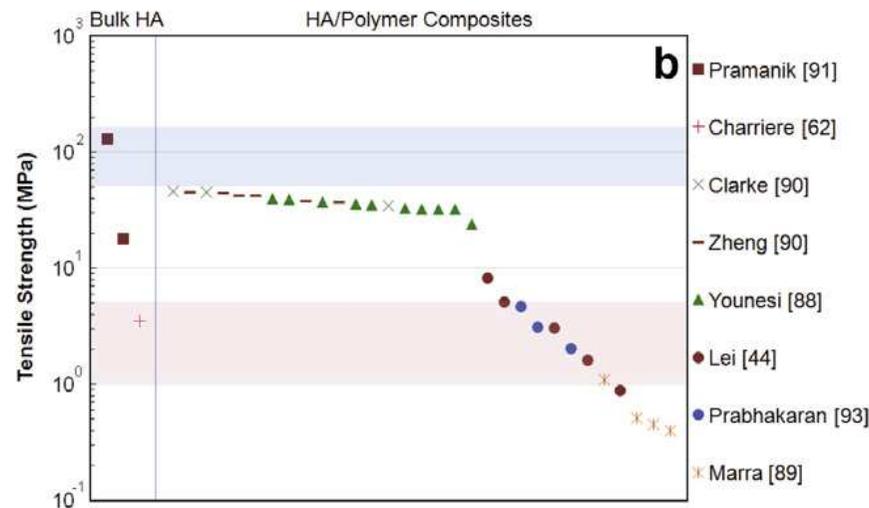
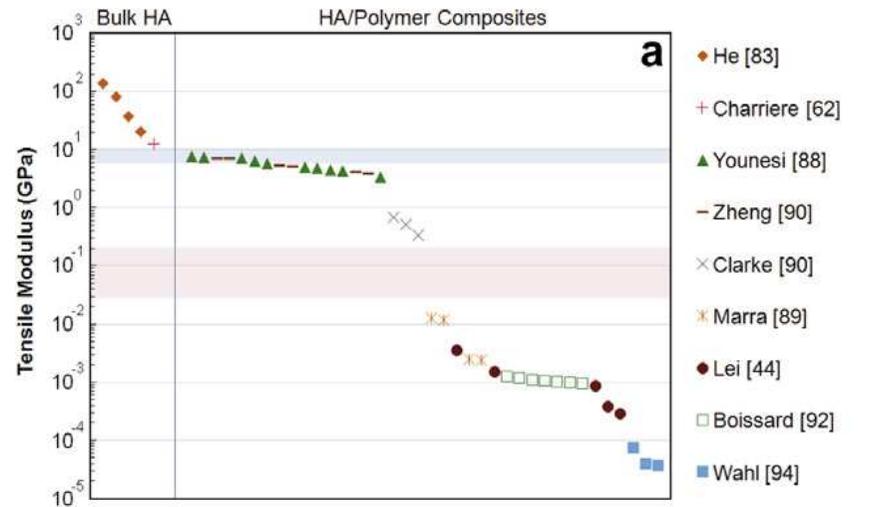
Bone Cement Mechanical & Fracture Properties

- Ceramic properties can be characterised as:
 - brittle (low fracture resistance, flaw tolerance)
 - low tensile strength (fibers are exception)
 - poor fatigue resistance (relates to flaw tolerance)
- Bone characterised as:
 - Composite in nature (>5 orders of magnitude in length scale)
 - Variable dependant on site and orientation
 - Viscoelastic
 - Poor fatigue resistance (but built-in repair mechanisms)
- **Implies ideal bone cement should be composite (Biomimetic)**

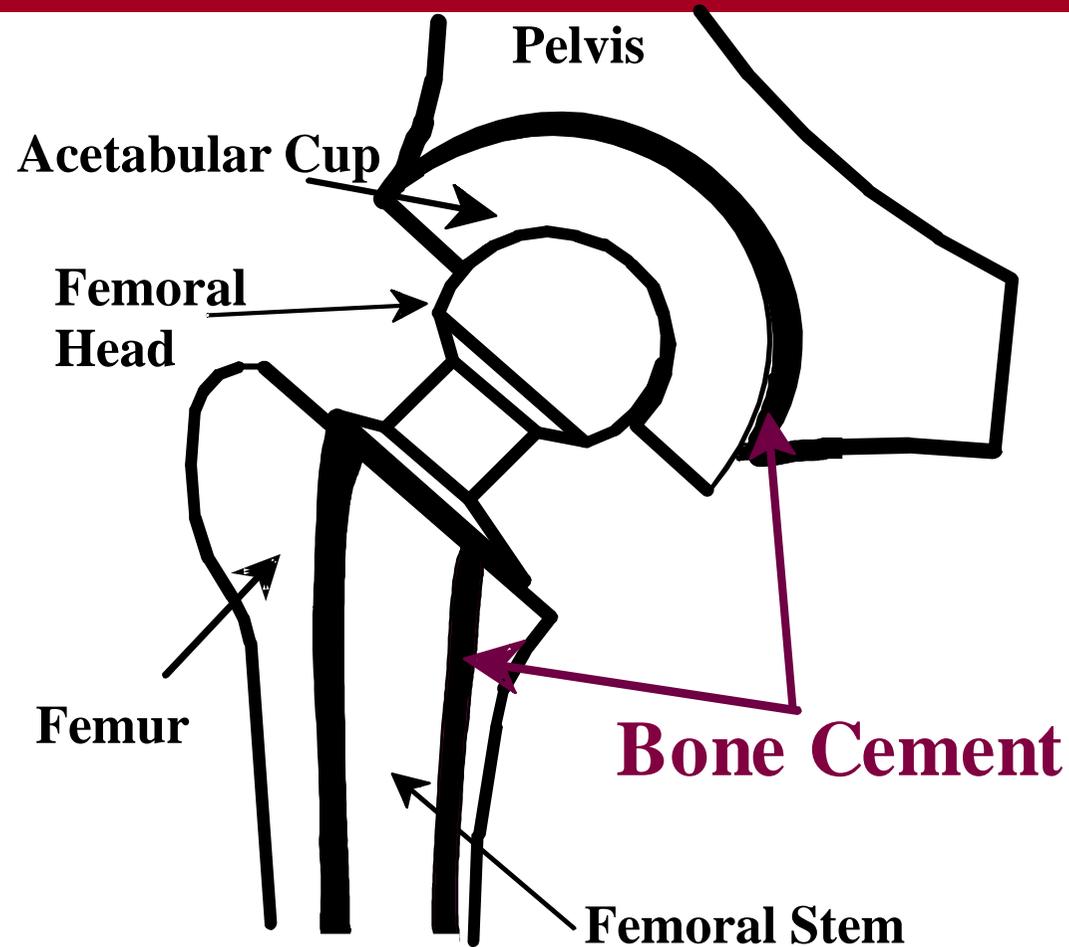
Compressive Strength of Bio-ceramics, Composites and Polymers ⁶



Tensile Modulus and Strength of Bio-ceramics, Composites and Polymers ⁶



BONE CEMENTS



in TOTAL JOINT REPLACEMENTS

PMMA Bone cement

2-component system: powder and liquid mixed 2:1

Powder

PMMA/PMA/PS spheres 30-150um (>90%)

Radiopacifiers (BaSO_4 / ZrO_2) (4-30%)

Initiator (benzoyl peroxide) (2-3%)

Liquid

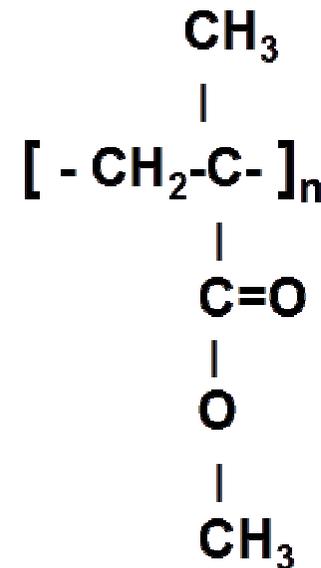
MMA monomer (>85%)

Co-monomers (10-15%)

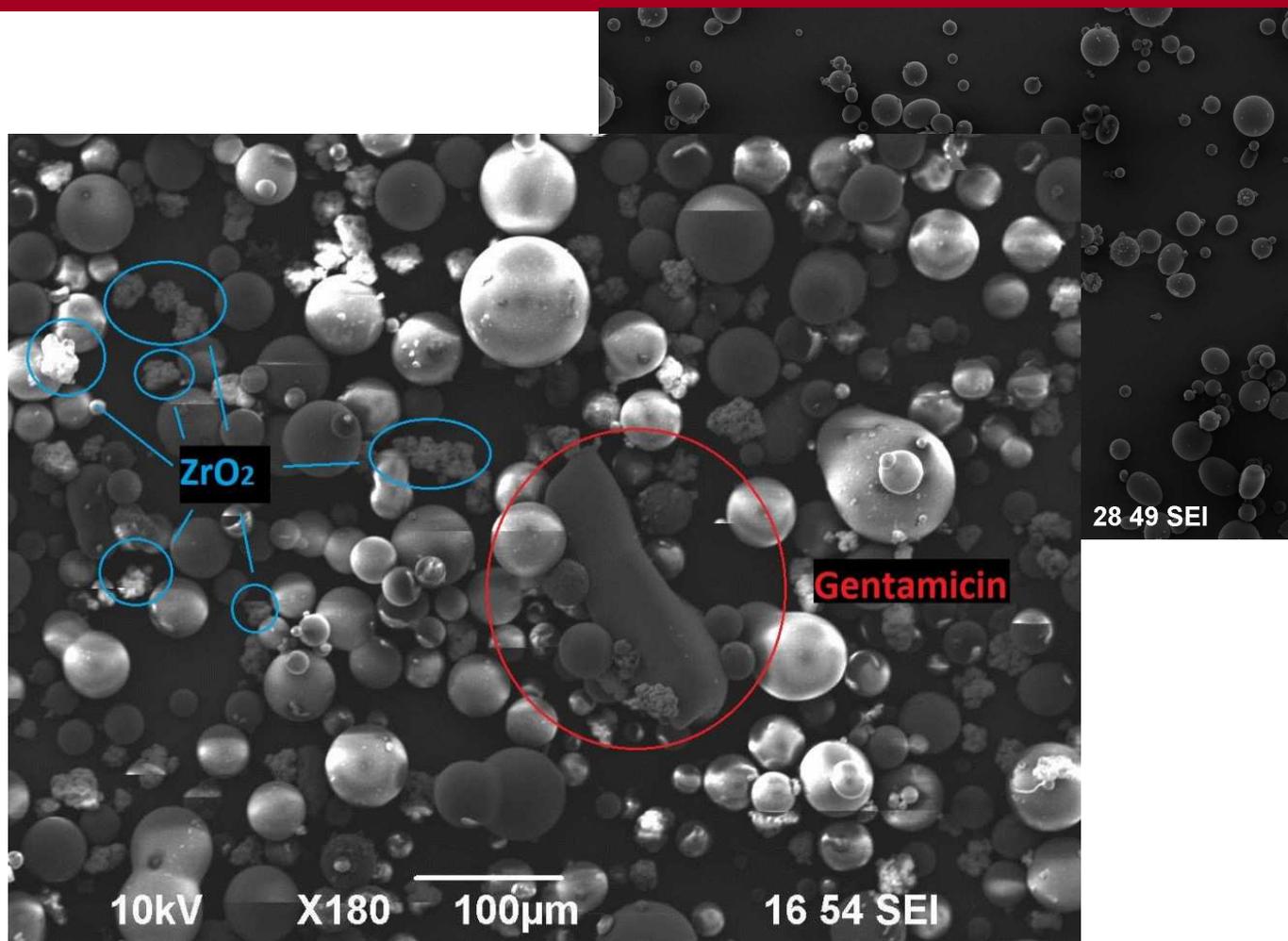
Inhibitor

Activator (Dimethyl-p –toluidine) (2-3%)

- Mix components together to a doughy stage
- Injected into prepared site, and allowed to cure via free radical polymerisation



Typical HV TKA Cement



PMMA Bone cement

Free Radical
Polymerisation

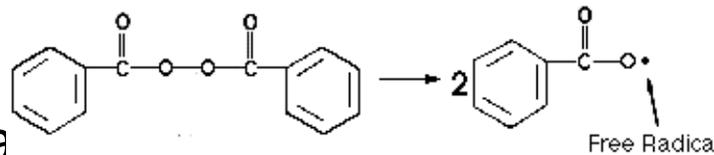
Initiation

Propagation

Termination

Initiation:

The powder and liquid components have been mixed
The DMPT breaks down BPO to produce 2 free radicals:

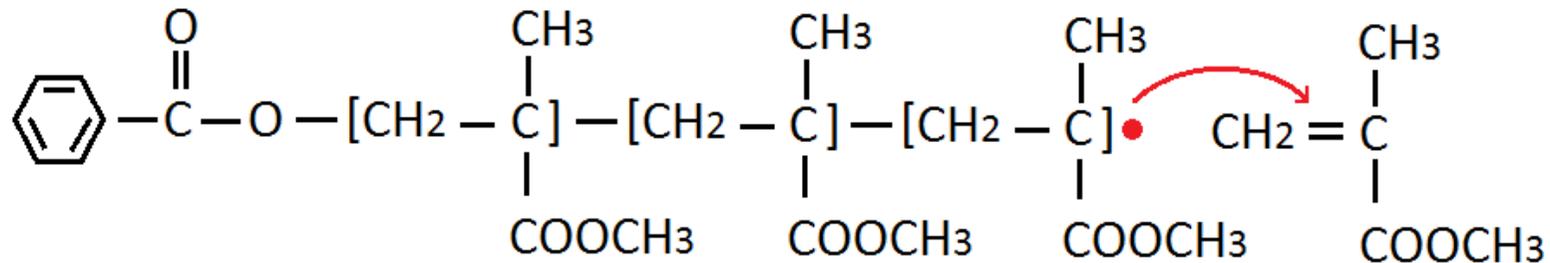


These free radicals at one end which denotes the active centre

This active centre is transferred to the end of the molecule as the polymer chain grows in size

PMMA Bone cement

Propagation:



Free radical attacks the C=C bond of the monomer, generating another radical

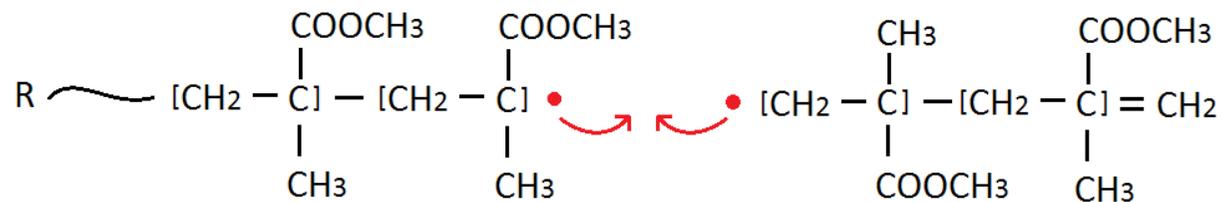
This radical attacks another molecule, adding to the growing polymer chain

This process continues growing the polymer chain until termination

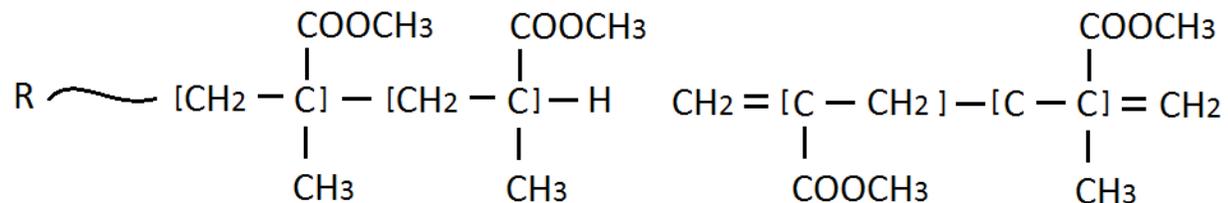
PMMA Bone cement

Termination:

- Combination:



- Disproportionation



PMMA Bone cement

Trommsdorff effect:

- An auto acceleration effect occurs at approximately 20-50% of conversion, due to an increase in viscosity
- In a more viscous medium, chain mobility is reduced, and the end chain radicals have a lower chance of being in a position to end termination
- However, small monomer molecules can still diffuse freely through the viscous medium to meet the active chain ends.
- As a result, there is a sharp increase in polymerization rate, which is accompanied by an increase in reaction exotherm
- This is why pre-polymer is added.

Causes for Revision (Mayo Clinic)

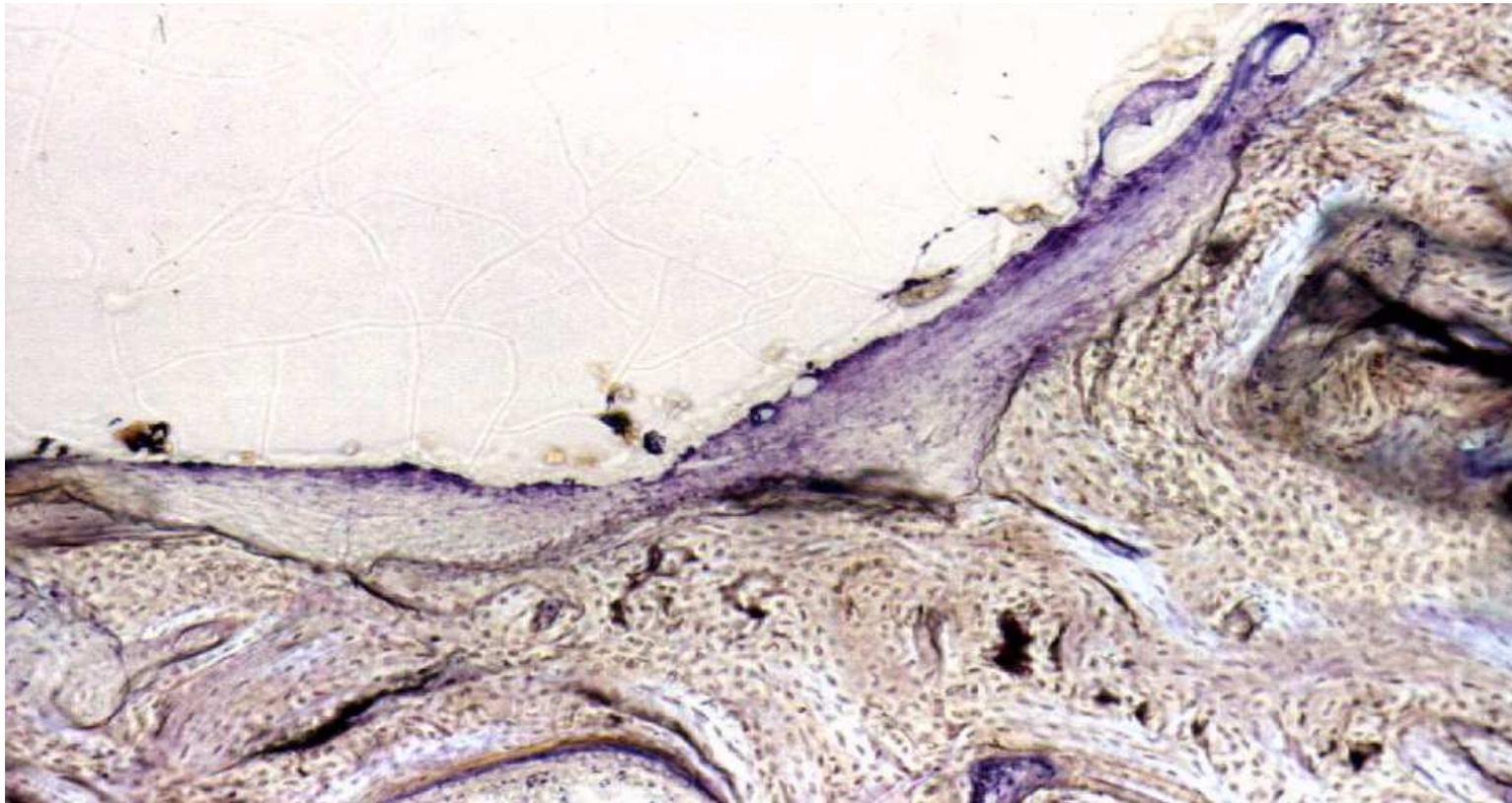
1. Aseptic Loosening
2. Fracture
3. Dislocation
4. Infection

Aseptic Loosening



Note the formation of a radiolucent layer as a result of fibrous capsule layer and stress shielding that leads to failure.

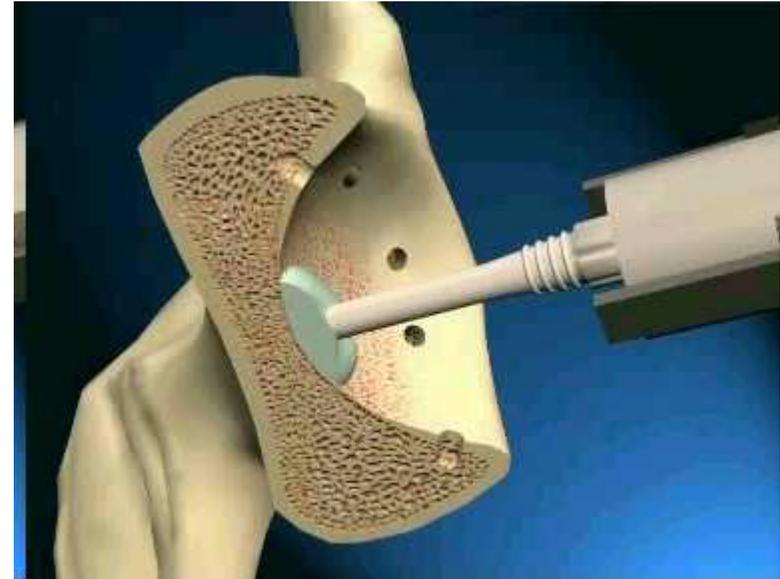
Fibrous Encapsulation



Fibrous reaction at the interface of PMMA and bone. A thickened, cellular, fibrous layer is present at the interface of the trabecular bone (Toluidine blue stained, ground section)

Contributions to Aseptic Loosening:

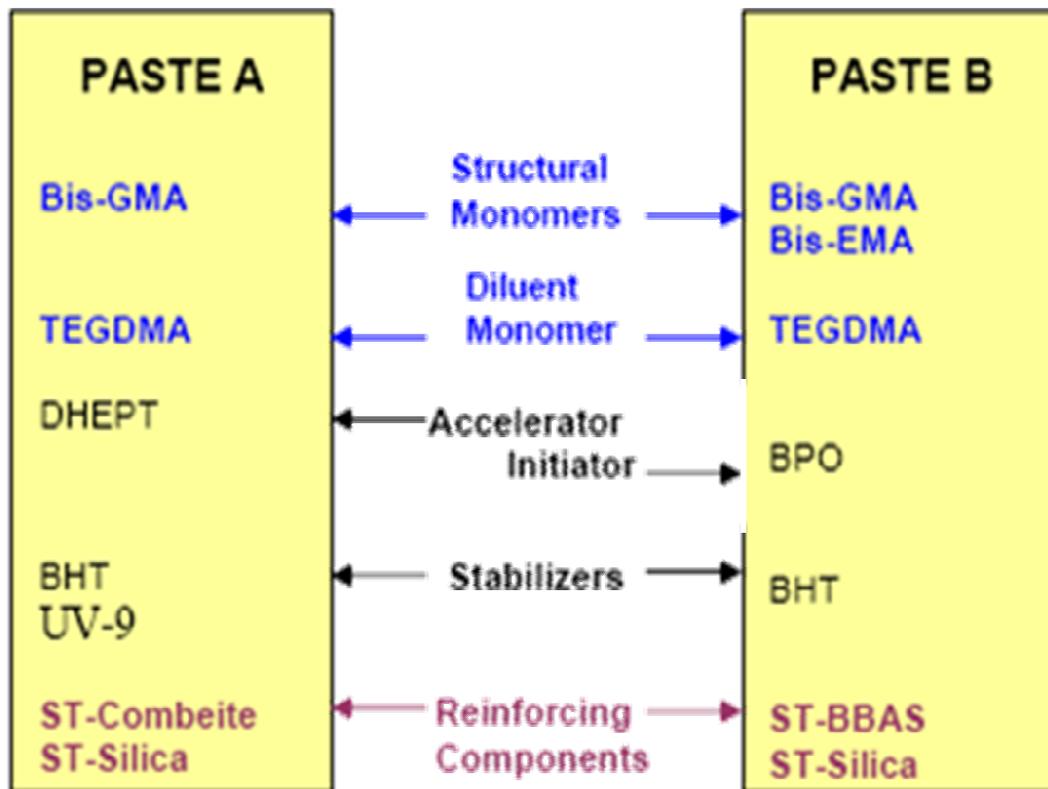
- Complex aetiology.
- No chemical bond.
- Thermal Necrosis.
- Chemical necrosis.
- Osteolysis
- Stress shielding –due to modulus mismatch.
- Shrinkage –up to 22%.
- Fibrous Capsule – Standard foreign body response



Properties Vs Design Criteria

Properties	PMMA bone cements
Quick setting (3 to 15 minutes)	Yes
Exotherm < 56° C	No
Matrix for drug delivery	No Intra operatively Yes
Osteointegrative (promotes bone growth)	No
Bioactive	if Bioactive ingredient added
Bioresorbable	No
Adequate viscosity	Yes
Radio-opaque	Yes if BaSO ₄ or ZrO ₂ is added
Modulus match with trabecular bone (~10-20 MPa)	No
Adhesive bond formation with bone/implant	No
Dimensional stability	No
<ul style="list-style-type: none">• Fibrous encapsulation• Local toxicity monomer• Systemic toxicity –cardiovascular, liver & immune impairment	

“Bioactive” Resin Composite- Cortoss™



Bioactive Resin Composite- Cortoss™

Di-functional Monomers

- bisGMA
- bisEMA
- TEGDM

Reinforcing / Bioactive Fillers

- 45S5 Glass Ceramic (Combeite crystalline phase)
- Barium-Borosilicate glass

Activator + Initiator

- BPO
- DMPT

Bioactive Resin VCF Composite- Cortoss™

Di-functional Monomers

- Cross-linked Network – increased stiffness
- high degree of monomer conversion, lower exotherm-63⁰C
- Higher Mw than MMA - decreased leachable toxic monomer

Reinforced with ceramic particles

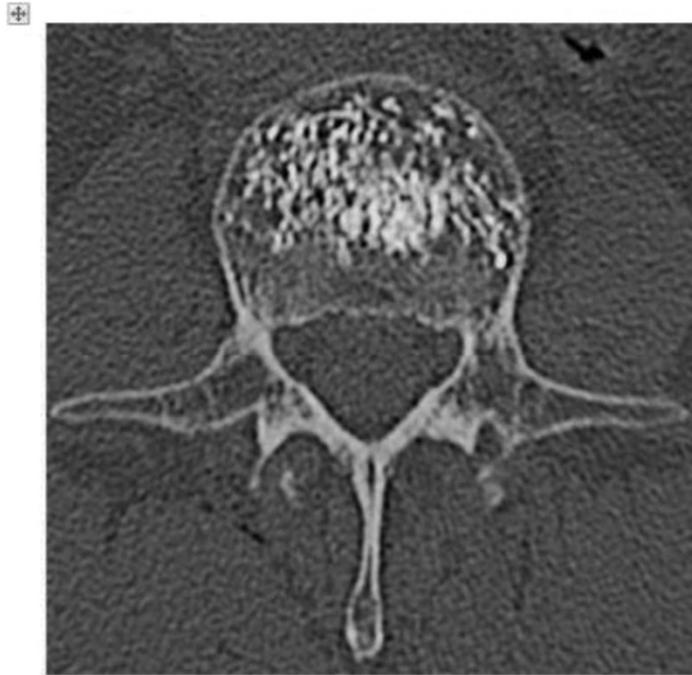
- increased stiffness

Bioactive Components

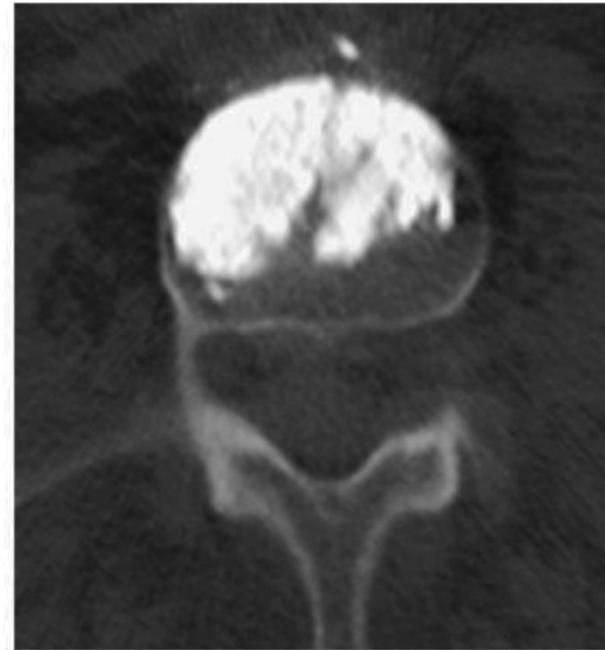
- Increased bone apposition at the interface
- Improved interfacial bond strengths between implant and bone

1. G.J. Pomrink, M.P. Diccico, T.D. Clineff, E.M. Erbe, *Biomaterials* 24, 1023 (2003)
2. E.M. Erbe, T.D. Clineff, G. Gualtieri, *Eur. Spine J.* 10, S147 (2001)

Cortoss™ Dispersed Fill of Vertebral Body



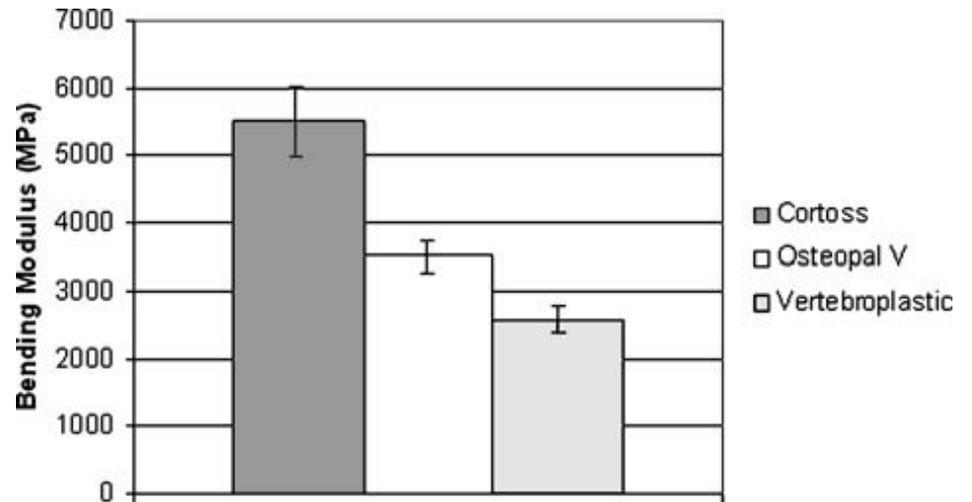
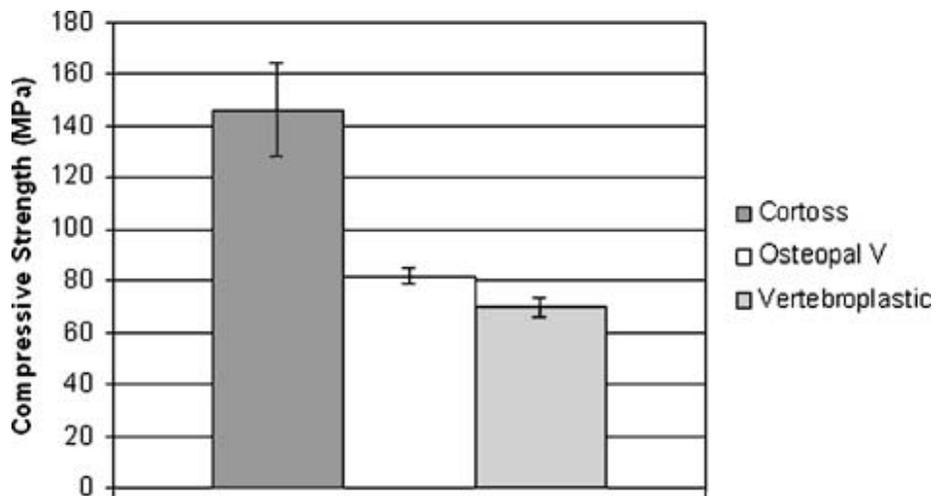
Cortoss



PMMA

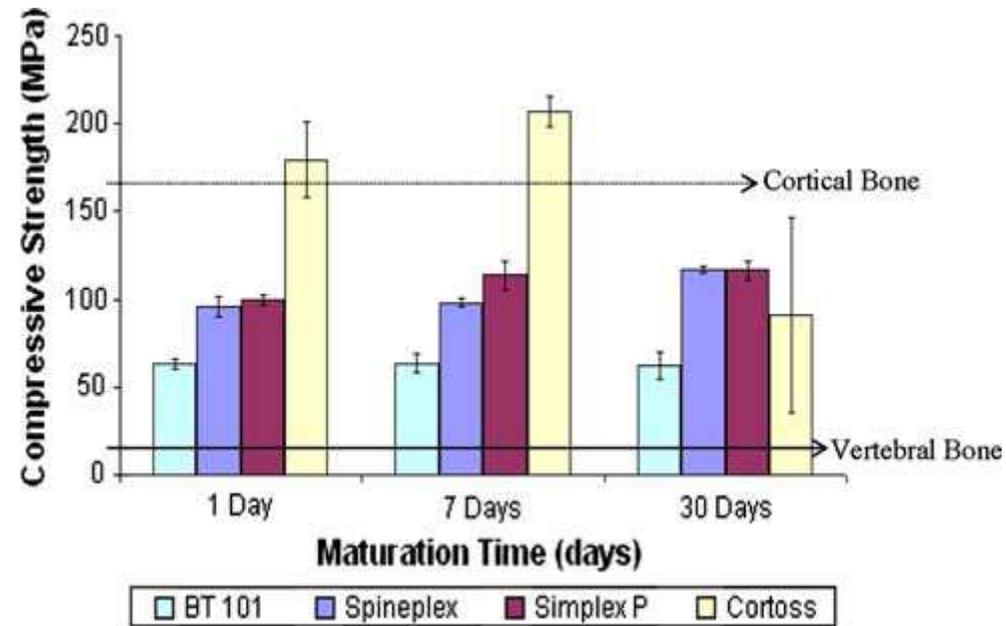
1. A Prospective Randomized FDA-IDE Trial Comparing Cortoss With PMMA for Vertebroplasty SPINE Volume 37, Number 7, pp 544–550 (2012)

Cortoss™ Mechanical Property Comparison



1. Sabina Gheduzzi · Jason J.C. Webb · Anthony W. Miles, J Mater Sci: Mater Med 17: 421–426 (2006)

Cortoss™ Strength Loss with Time

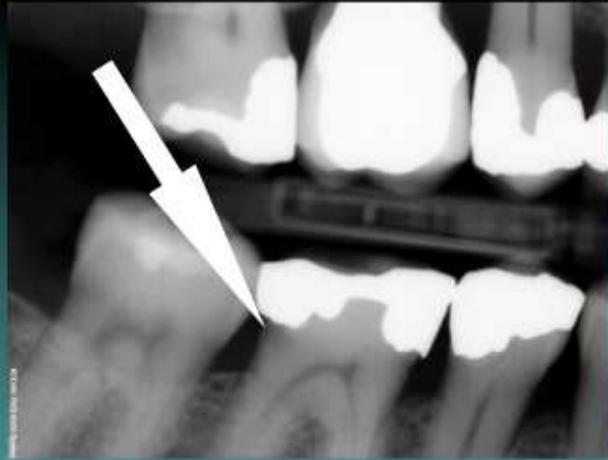


1. D. Boyd , M. R. Towler , A. Wren , O. M. Clarkin J Mater Sci: Mater Med 19:1745–1752(2008)

Bioactive Resin VCF Composite- Cortoss™

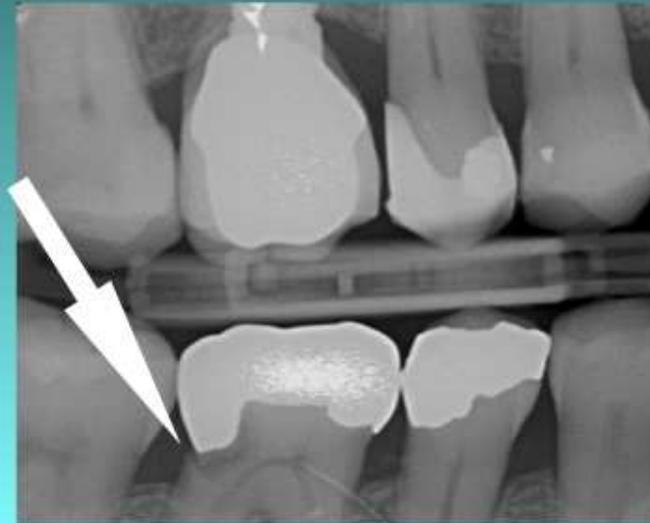
Disadvantages

- Exotherm- 63°C
- Cold Chain Product
- bisEMA more prone to water uptake
 - Plasticisation of matrix – loss of stiffness
 - Solubility of large Bioglass particles results in swelling and micro-cracking of matrix – loss of strength
- Bioactivity modest given majority of Bioglass is in crystallised form
- Brittle $K_{IC} \sim 0.7 \text{ MPa}\sqrt{\text{m}}$



Cavity under
amalgam restoration

After crown preparation
more caries developed





Temporary crown filled with Bio-active glass, mineralizes decayed dentine, respecting its vitality!

Bioactive crown cemented with bioactive glass, creating a fusion between the tooth and the crown of hydroxy/fluor apatite



Can a BisGMA Composite be made REALLY Bioactive?

- Yes
- Incorporate BAG of greater bioactivity than 45S5...

BAG Filed BisGMA Composite QMUL 2016

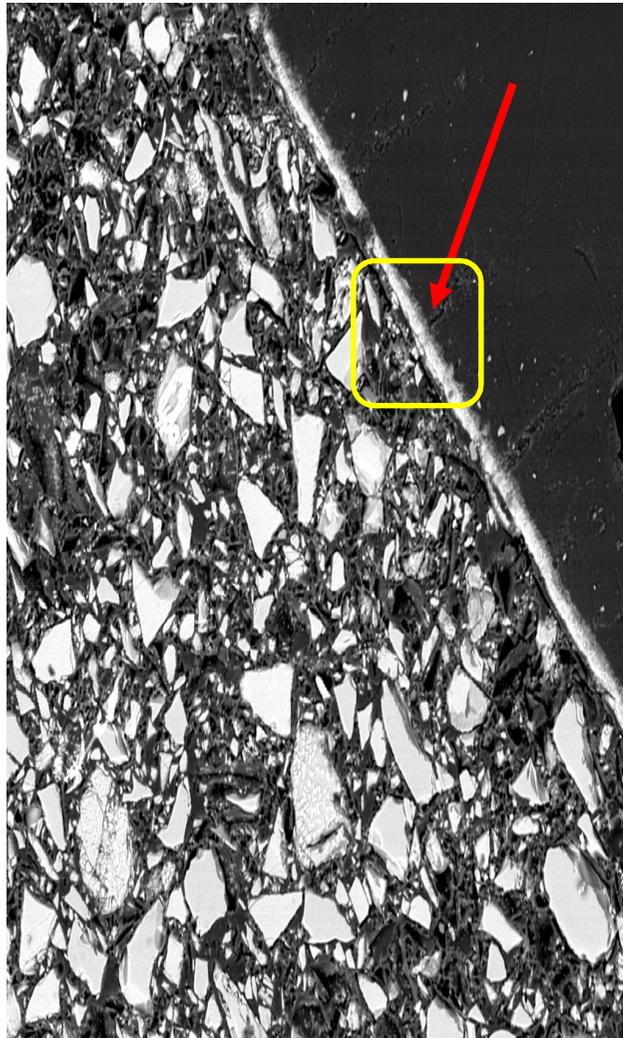
Verify via

- Age in
 - SBF / Artificial Saliva / TRIS Buffer
 - Static – Assess Ca P Deposition (volumetric)
 - Dynamic – Assess Solution Concentration (Flux) in sink conditions

NB Ionic Strength

- FTIR
- XRD
- MAS NMR

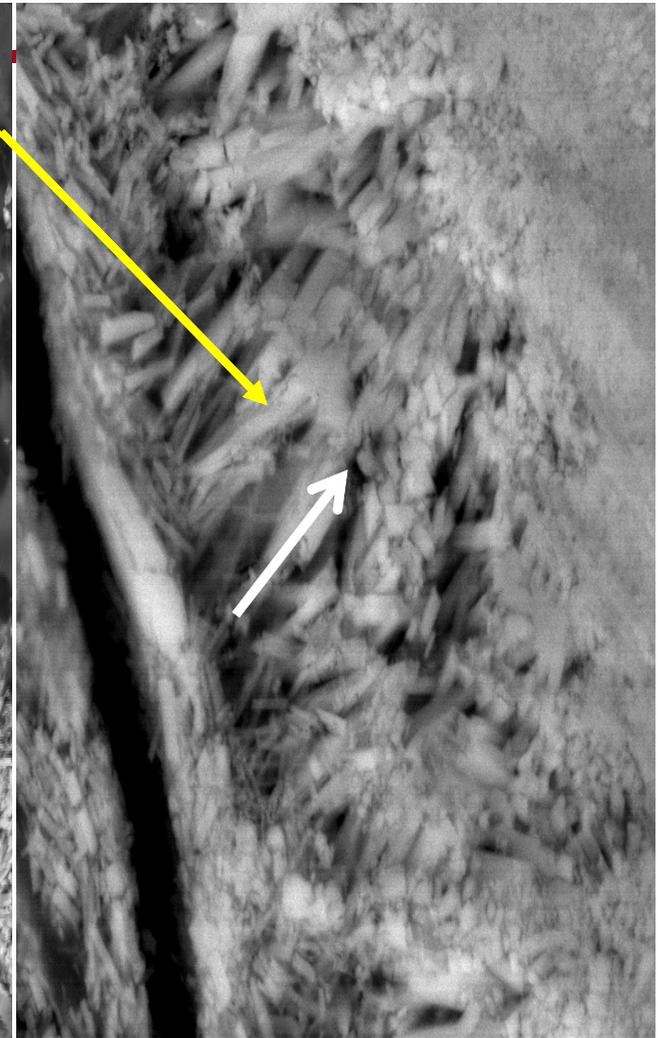
BAG Filed BisGMA Composite QMUL 2016



14/12/2015 HV mag □ WD
12:08:09 10.00 kV 1 000 x 10.0 mm
Queen Mary University of London

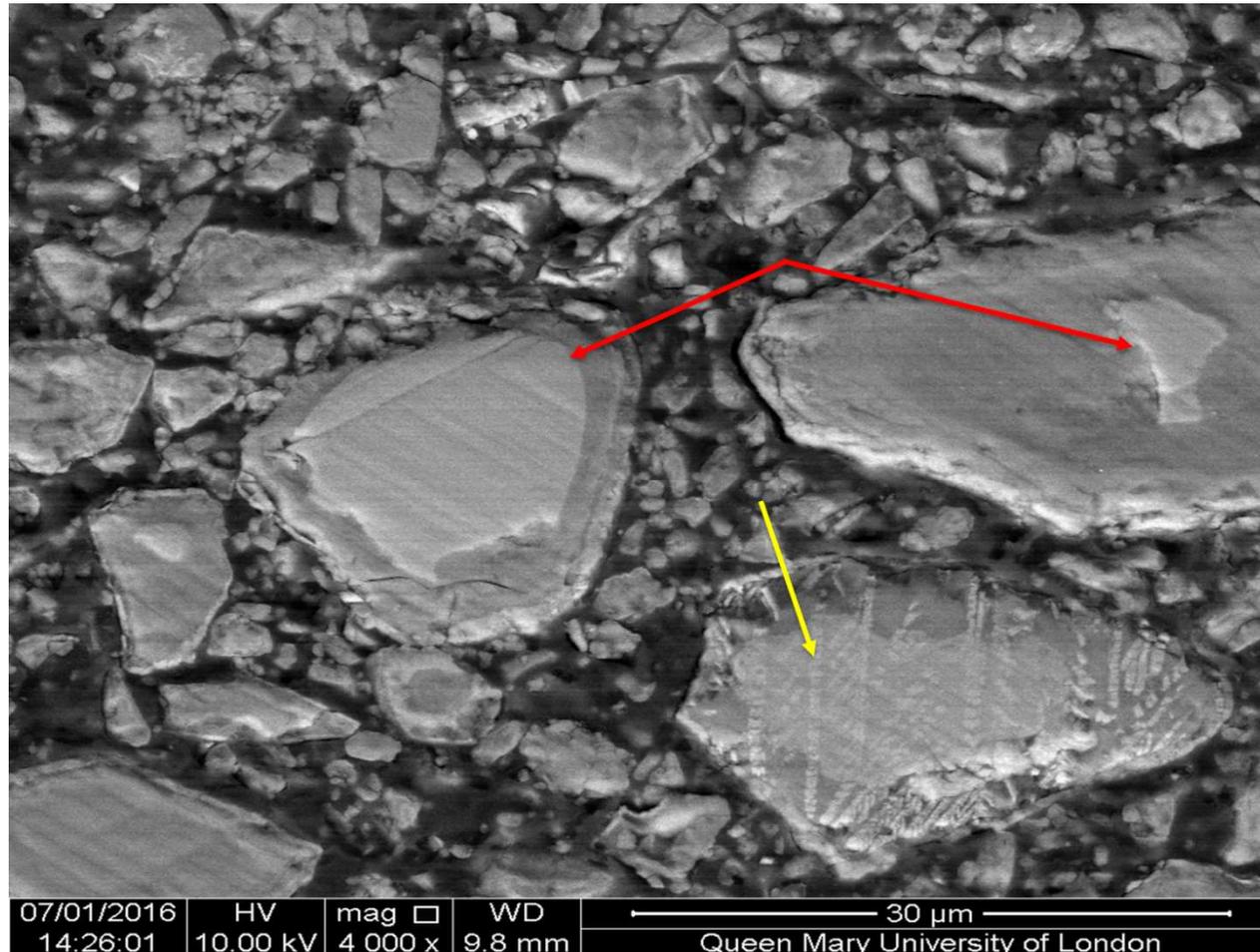


14/12/2015 HV mag □ WD
12:12:40 10.00 kV 10 000 x 9.9 mm
Queen Mary University of London



05/01/2016 HV mag □ WD
12:36:24 10.00 kV 40 000 x 10.6 mm
Queen Mary University of London

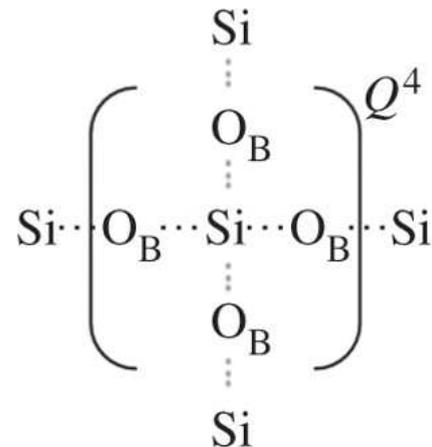
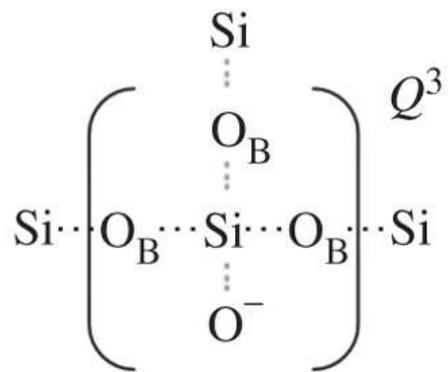
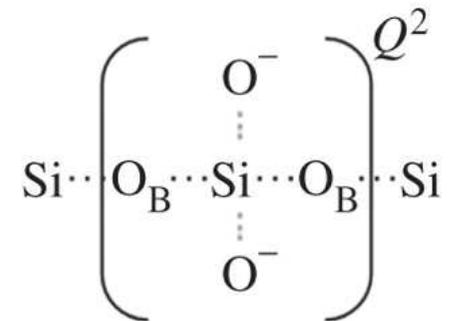
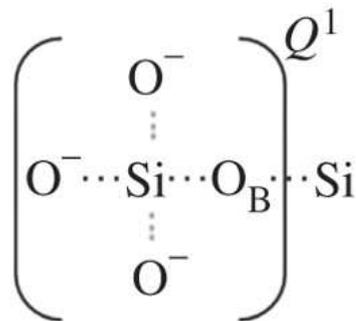
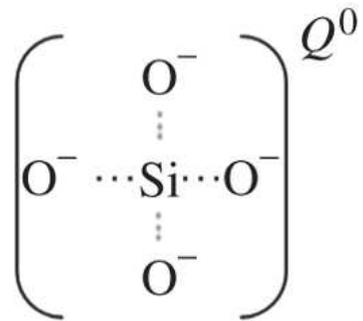
BAG Filed BisGMA Composite QMUL 2016



Glass Structure

- **Glasses comprise:**
- Network Formers. Form the 3D backbone of the glass. Eg SiO_2 .
- Network Modifying Oxides. Break up the glass network eg Na_2O , K_2O , CaO , SrO .
- Bridging Oxygens (Si-O-Si).
- Non-Bridging Oxygens $\text{Si-O}^- \text{Na}^+$.

Schematic of the different Q structures that can describe Si network connectivity in glasses.



O_B represents a network-forming bridging oxygen

Network Connectivity

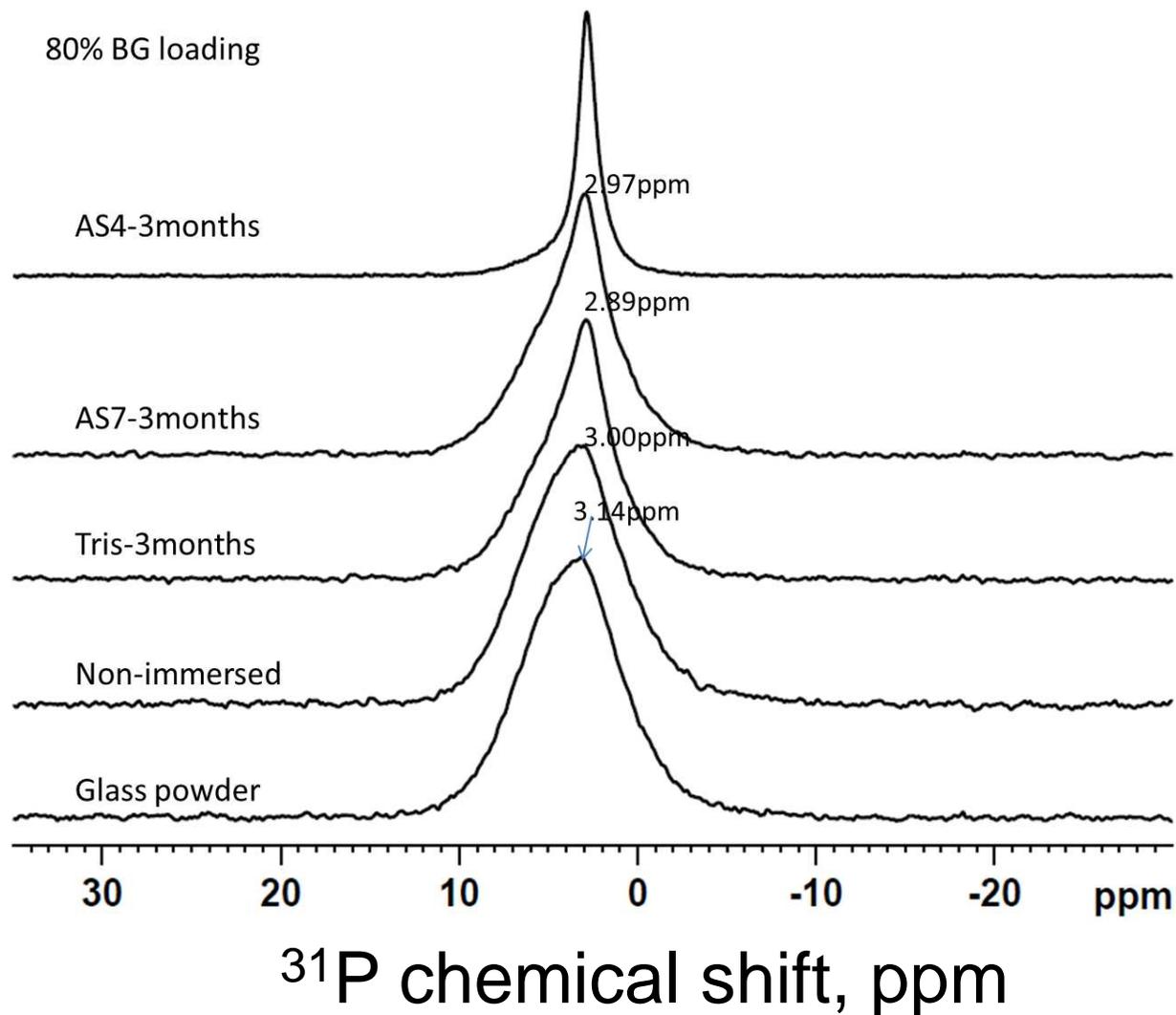
$$NC = 2 + \frac{[(2 \times SiO_2) + (2 \times P_2O_5)] - [2(M_2^I O + M_2^{II} O)]}{SiO_2 + (2 \times P_2O_5)}$$

The formula above assumes that P atoms are in network forming role in a range of Q^n structures, and the version below assumes that P is an orthophosphate Q^0 structure and requires modifier cations in a charge balancing role

$$NC = \frac{4[SiO_2] - 2[M_2^I O + M_2^{II} O] + 6[P_2O_5]}{[SiO_2]}$$

where $M_2^I O$ and $M_2^{II} O$ are the mono- and divalent modifier oxides in the glass

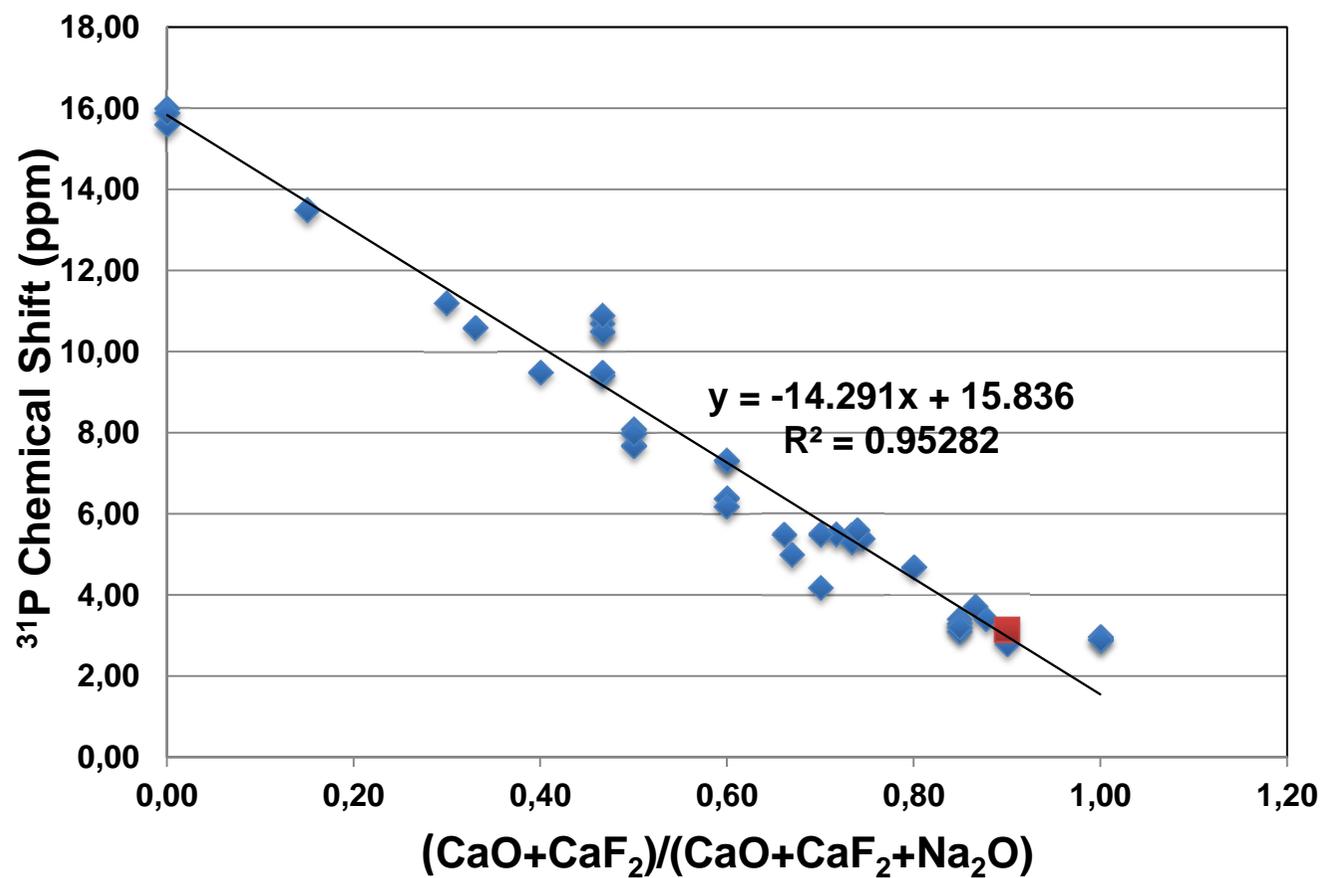
BAG Filled BisGMA Composite ^{31}P MAS NMR



BAG Filed BisGMA Composite QMUL 2016

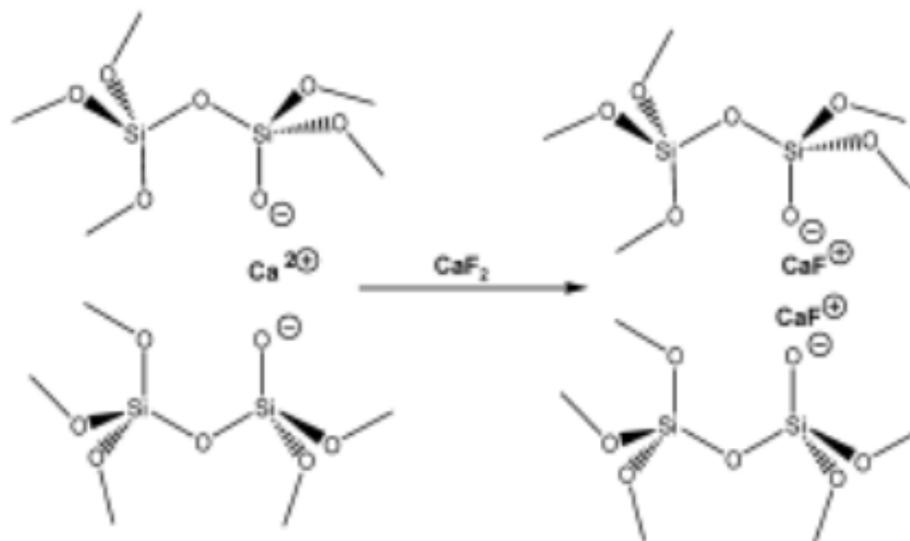
- The ^{31}P MAS-NMR spectra of the glass and the non immersed composite exhibit a broad peak at approximately 3ppm corresponding to a mixed Q^0 orthophosphate PO_4^{3-} species charge balanced by Ca and a small amount of Na.
- With increasing immersion time a sharp peak at 2.8-2.9ppm develops corresponding to apatite and increases with time with the broad signal from the original glass simultaneously disappearing.
- By 3 Months there ^{31}P MAS-NMR signal corresponds to only apatite indicating that **all** the phosphorous content of the glass has been consumed and has been **converted to apatite** within the composite disc. Bulk (as opposed to surface) assessment of reactivity
- The charge balancing of the orthophosphate would be expected to correlate with the $\text{CaO}/(\text{CaO}+\text{Na}_2\text{O})$ ratio of the glass composition

Predicted ^{31}P MAS NMR Chemical Shift

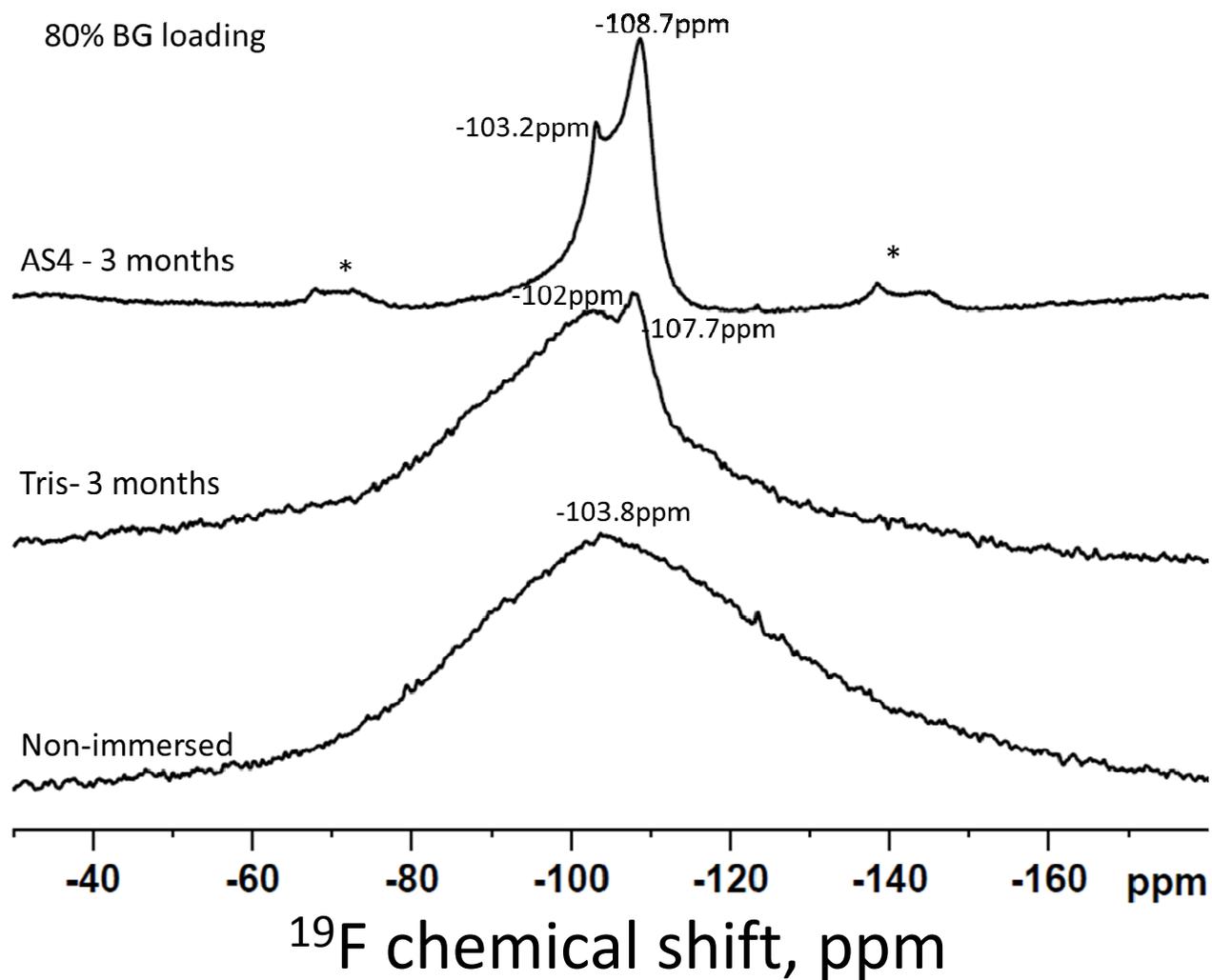


Example Ionomer Glass -Sodium Series

Peak	Species	Chemical shift, ppm	Percentage
1	F-Ca(n)	-97.7	44.9
2	F-Ca(3)Na(1)	-130.2	7.6
3	Al-F-Ca(n)	-158.3	37.8
4	Al-F-Na(n)	-189.0	9.6



BAG Filled BisGMA Composite ^{19}F MAS NMR

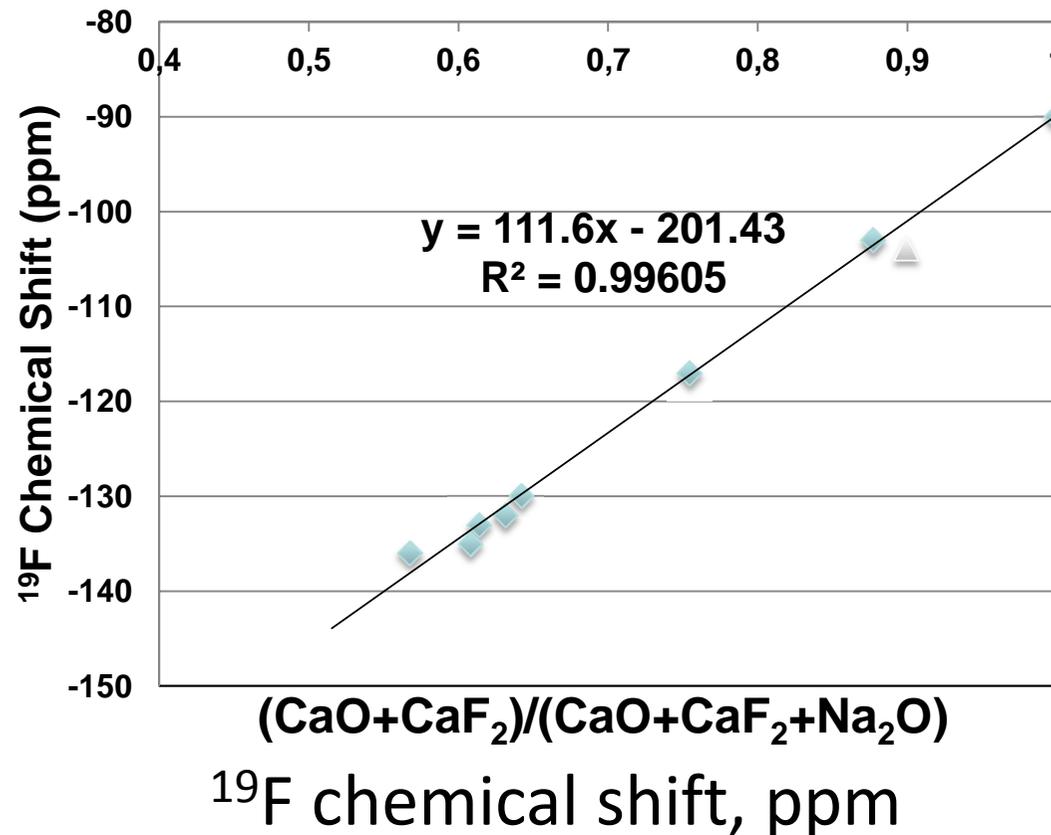


BAG Filed BisGMA Composite QMUL 2016

- F was complexed by Ca in a sodium free and the
- dominant species present F-Ca(n).
- Given the low Na₂O content of the glasses the dominant species are likely to be F-Ca(n) species where n is 3/4.
- The ¹⁹F Chemical shift for an F-Ca(n) species is about -100ppm and the peak exhibited a maximum at a chemical shift of -103ppm.
- With increasing immersion time the broad peak from the original glass disappears and two new sharper peaks at -103 and -108ppm develop corresponding to fluorapatite(FAP) and calcium fluoride (CaF₂). There is probably a third peak present at -105ppm corresponding to possibly a mixed fluorohydroxyapatite. By 3 Months the broad signal from the original glass has been lost indicating that all the original fluorine in the glass has been consumed in forming largely FAP and CaF₂.
- The formation of CaF₂ is considered undesirable and indicates the fluoride content of the glass is too high to be ideal.

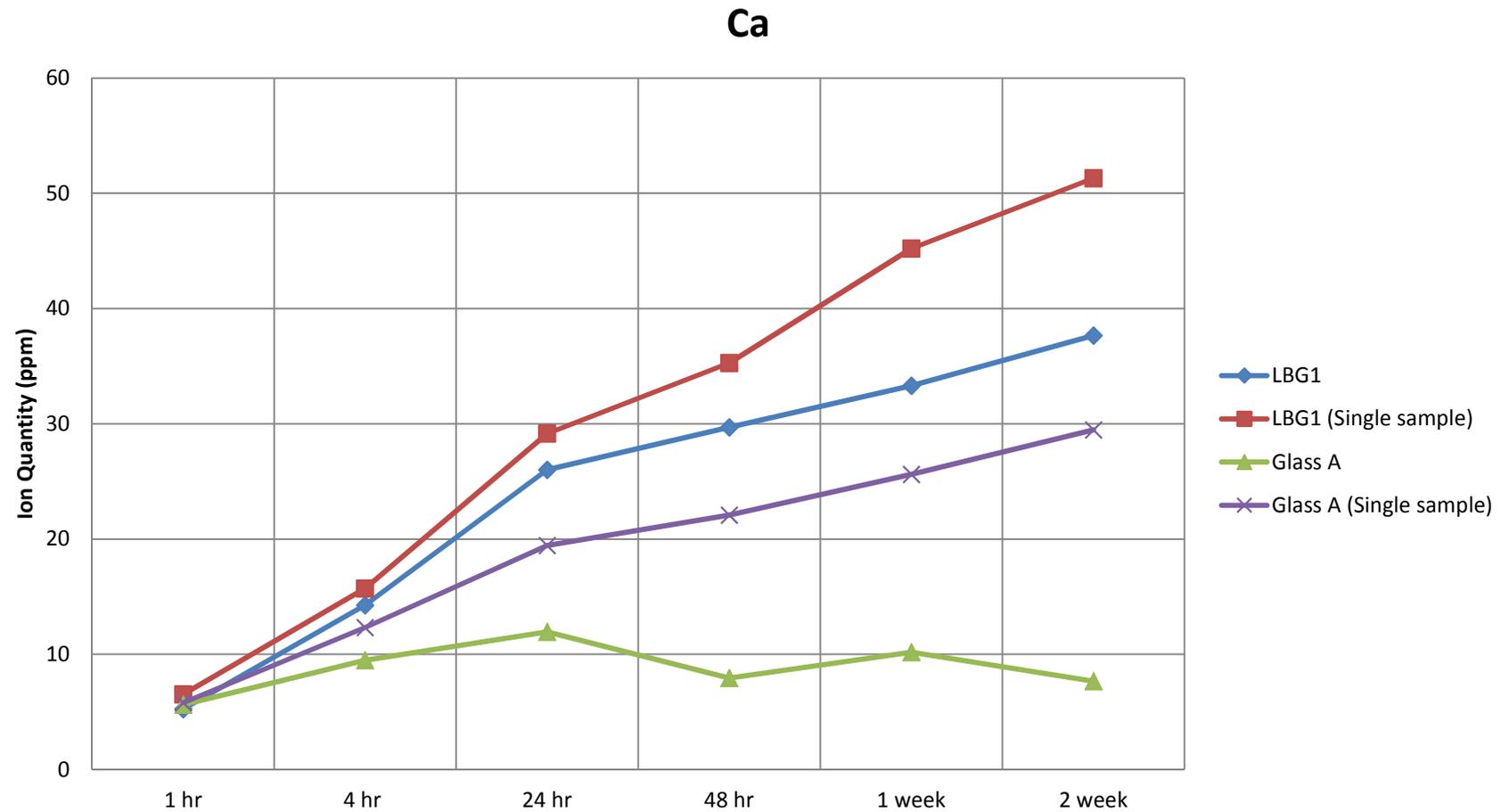
^{19}F Chemical Shift for Glass

- Prediction based on Ca/Na ratio -100ppm.
- Based on ^{31}P chemical Shift -87ppm.



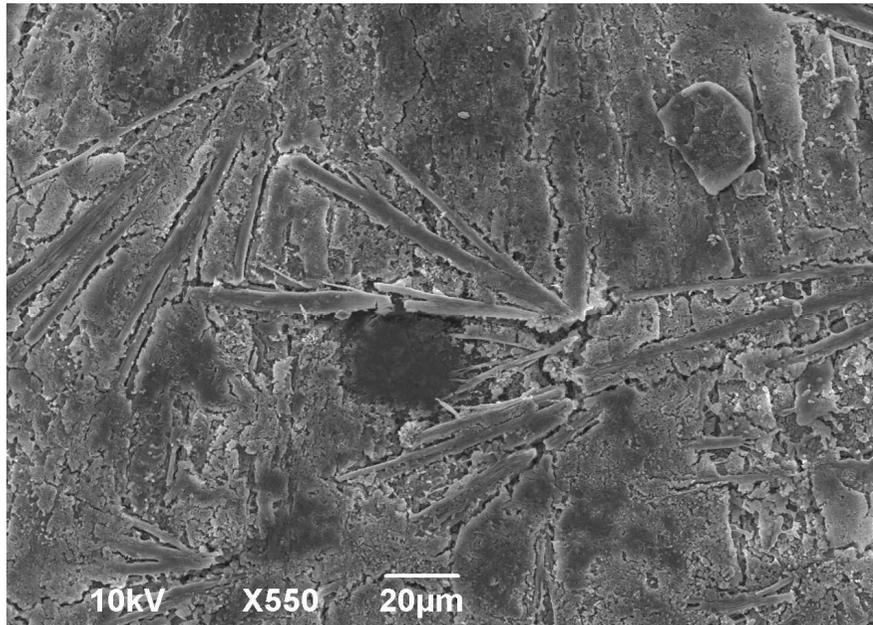
PMMA Bone Cement be made REALLY Bioactive?

Conventional PMMA Cement + BAG

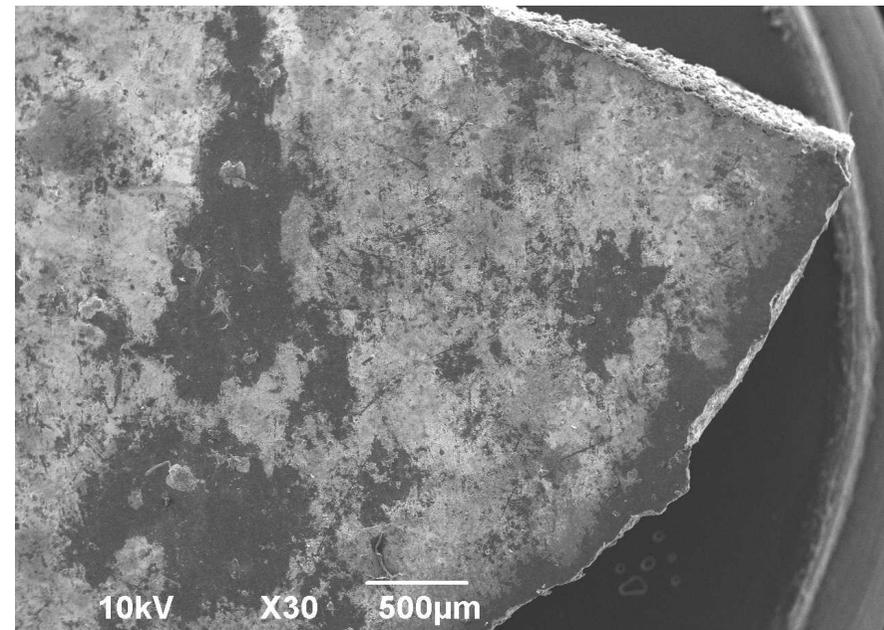


TRIS Buffer Dynamic @37 C

Conventional PMMA Cement + BAG

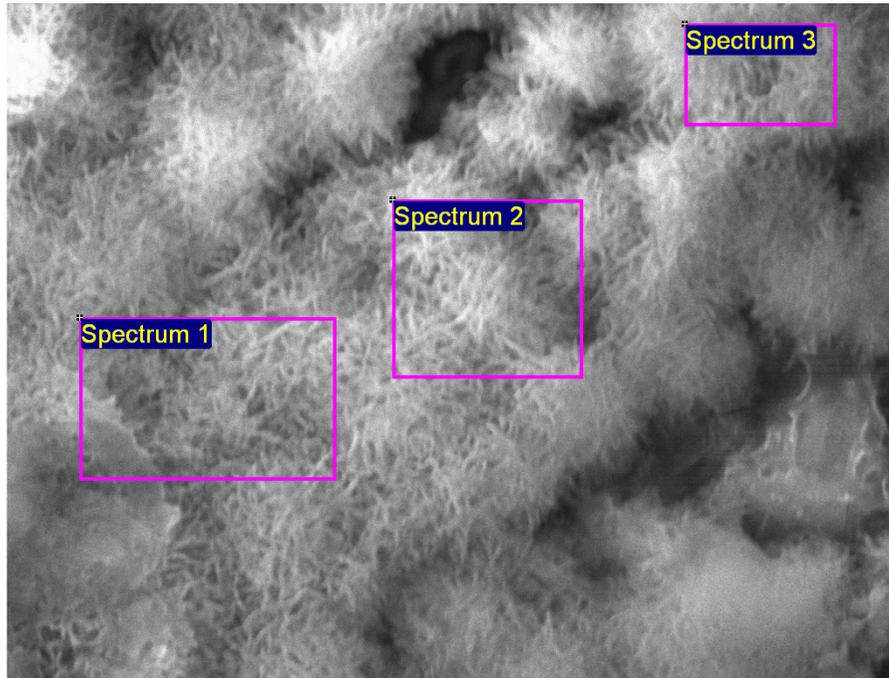


TRIS Buffer 6 Hours Static



Conventional PMMA Cement + BAG

TRIS Buffer 6 Hours Static



3µm

Electron Image 1

Spectrum	In stats.	Na	Si	P	Ca	Total	
Spectrum 1	Yes	7.28	2.78	29.82	60.12	100.00	
Spectrum 2	Yes	8.65	7.20	28.62	55.53	100.00	
Spectrum 3	Yes	8.90	13.37	27.12	50.61	100.00	
Mean		8.28	7.78	28.52	55.42	100.00	
Std. deviation		0.87	5.32	1.35	4.75		
Max.		8.90	13.37	29.82	60.12		
Min.		7.28	2.78	27.12	50.61		

Predicting the bioactivity of glasses using the network connectivity or split network models

Conclusion:

The network polymerisation (Q^n structure) strongly influences glass dissolution and subsequent apatite formation, and the NC or split network models are useful and successful in predicting bioactivity

- i) They do not take account of the nature of the network modifying cations, in particular their charge to size ratio and their influence on the glass network.
- ii) They equate glass dissolution directly to bioactivity.

Predicting the “bioactivity” of Cements

Conclusion:

Can a cement be too bioactive?

Yes – in vivo mineralisation of :

- Adjacent soft tissue (!)
- Adjacent neural tissue (!!!!)

Network Connectivity Models can predict dissolution

Which in turn

Can be used to design materials exhibiting varying flux of ions which should be assessed for actual bioactivity IN VIVO

Predicting the “bioactivity” of Cements

Conclusion:

Network Connectivity Models can predict dissolution

Structural role of modifier and intermediate cations depends on type, ratio and quantity of other cations

Bioactivity assessment best done via total quantity or total flux of ions in entire system, informed by chemical equilibrium of that system

Bioactivity not generalised characteristic – should be tailored for specific indication of anatomical site and defect size

Acknowledgements

Dr Ian Houlihan

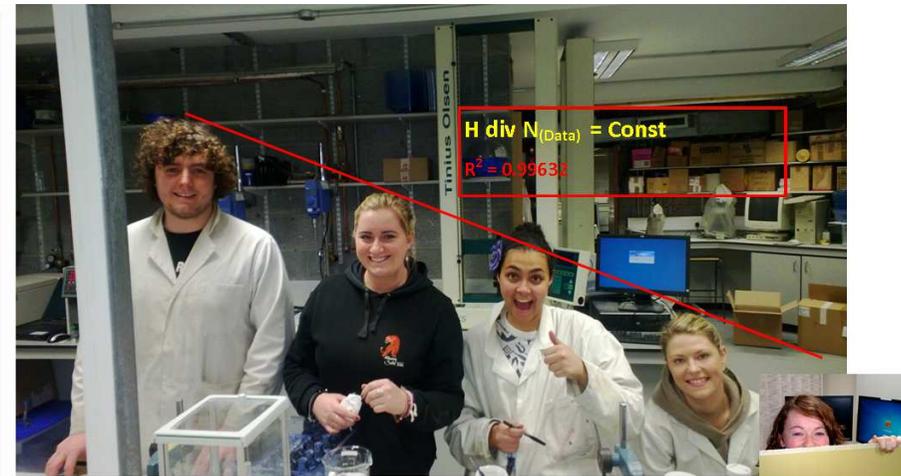
UL

Dr Mai Huang

QMUL

SS- MAS NMR

- Prof Robert Hill QMUL
- Dr. Natalia Karpukhina QMUL
- Dr Rob V. Law Imperial



1 / Quantity of Data

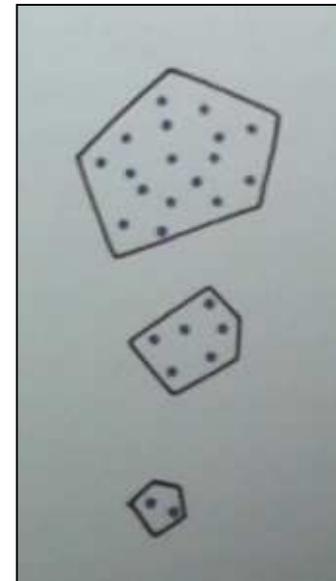
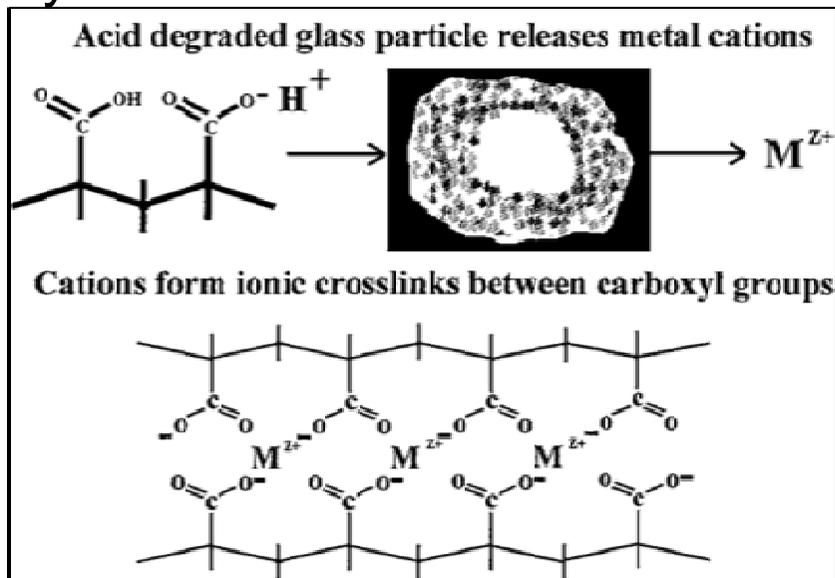
Conclusion: Team performance per metre is a constant



stryker®

Glass Ionomer Cements

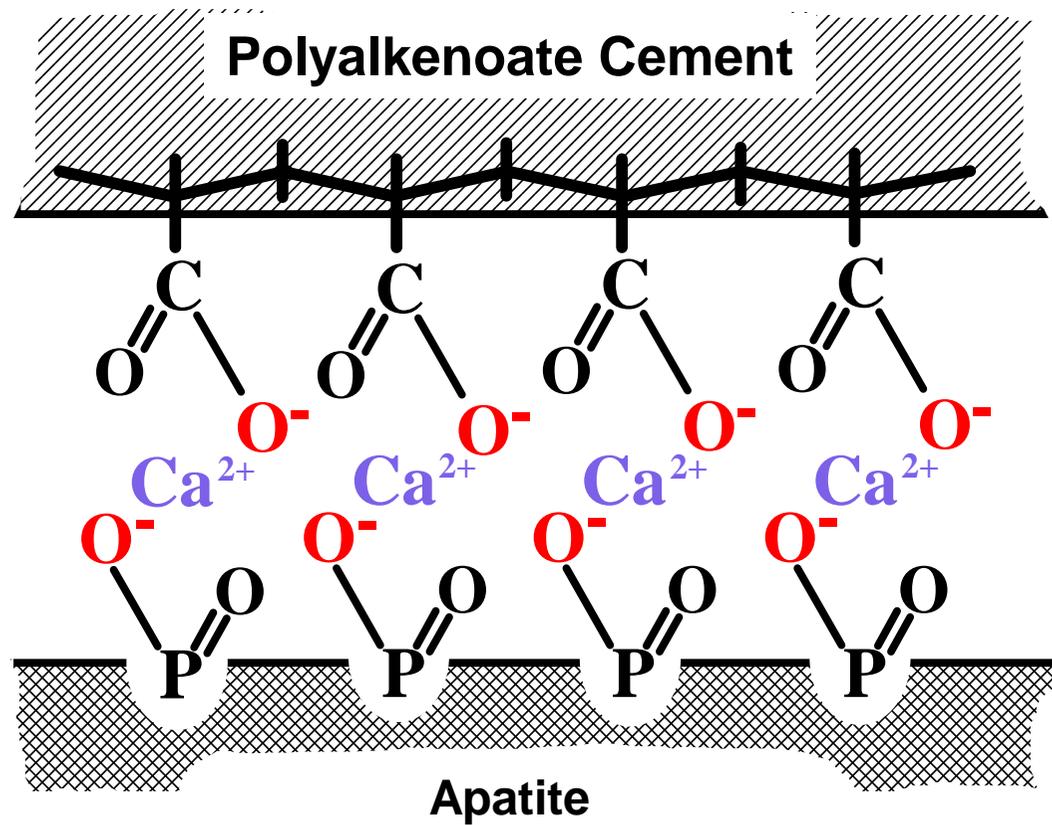
- Bioactive cement system
- Ion leachable glass powder, poly (acrylic) acid and water
- Carboxylic acid hydrolyses and degrades the glass, releasing ions
- Ions are chelated with COO^- groups
- Crosslinking of the polyacrylate chains, embedding glass particles in polysalt matrix



Ionomer Glasses

- Fluoro-alumino-silicate glass
- Glass is designed to contain a similar ratio of Ca–P cations for bone formation (If heat-treated crystallises to apatite and mullite)
- Therapeutic release of fluoride anions by cement reduces the risk of secondary caries (dentistry) and stimulates bone deposition (orthopaedics)

Adhesion



Comparison between GIC and PMMA Cements

GIC	PMMA
<ul style="list-style-type: none">• Chemically adhere to bone and dentin	<ul style="list-style-type: none">• Mechanically adhere to bone
<ul style="list-style-type: none">• No exotherms on setting	<ul style="list-style-type: none">• Thermal necrosis of tissue due to large exotherms on setting
<ul style="list-style-type: none">• No shrinkage of the cement	<ul style="list-style-type: none">• Shrinkage due to polymerisation
<ul style="list-style-type: none">• Therapeutic release of ions such as fluoride	<ul style="list-style-type: none">• Chemical necrosis due to leaching of monomer

Adhesion

- Chemically bond to dentin and bone
- Formation of chemical complexes to substrate
i.e. dentine, enamel, cortical and cancellous bone
- Bond to both the organic (collagen) and inorganic (apatite) components
- Adheres to any coherent oxide/passivating layer
ie TiO_2 , Cr_2O_3 , NiO

Adhesion



GPA cement applications:

- (1) restorative tooth filling materials
- (2) luting cements (adhesives)
- (3) Otological devices
- (4) Crannio/Maxillofacial reconstruction**
- (5) Alveolar ridge enhancement

GPA cements have the potential to be used as bone substitutes and cements

Deficiencies of Commercial GICs

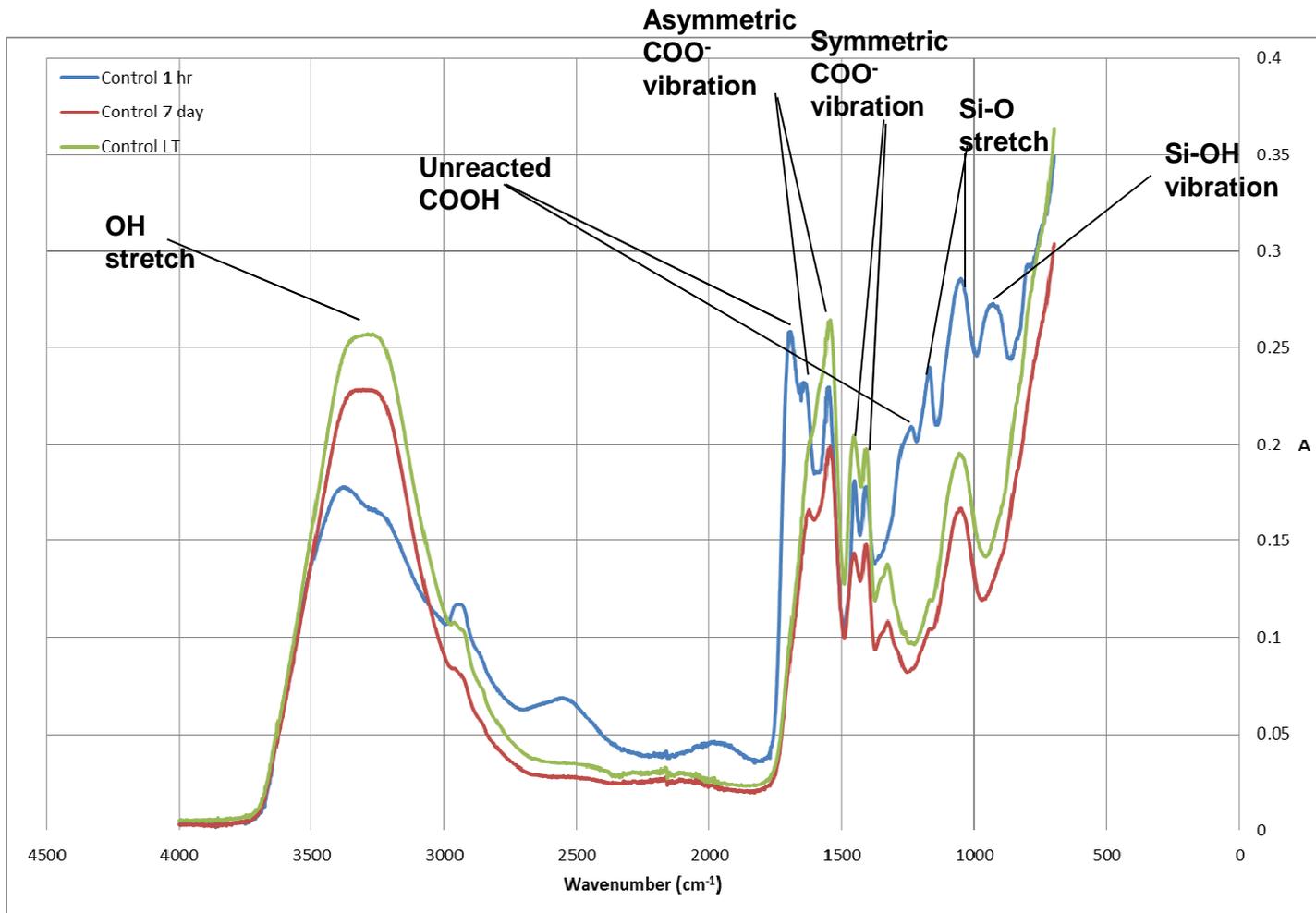
- Brittle material, lacks the toughness and fracture toughness for high load bearing applications

	Bone	GIC Commercial
<i>Modulus</i>	7 – 20 GPa	8 – 10 GPa
<i>Toughness</i>	1500 J/m ²	150 J/m ²
<i>Fracture Toughness</i>	0.5 – 2.2 MPa√m	0.5 – 0.9 MPa√m

Factors effecting GIC mechanical properties

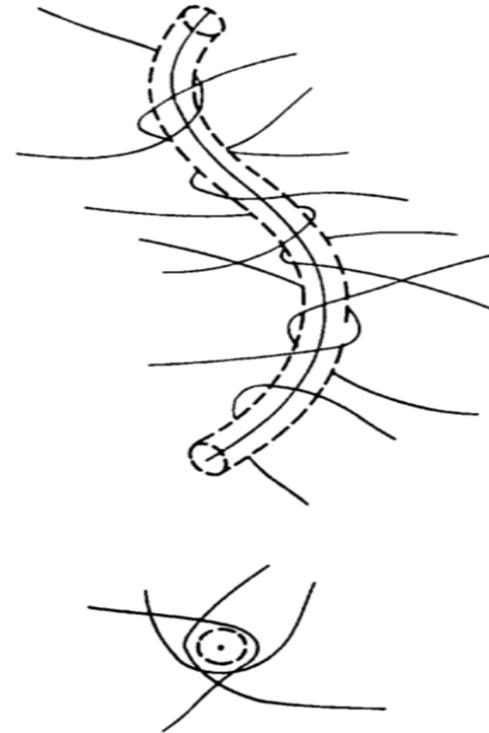
- Glass reactivity and composition
- PAA molecular weight
- Use of copolymers
- Particle size and morphology of powders
- Conditioning the glass particles

FTIR



Fracture of GICs

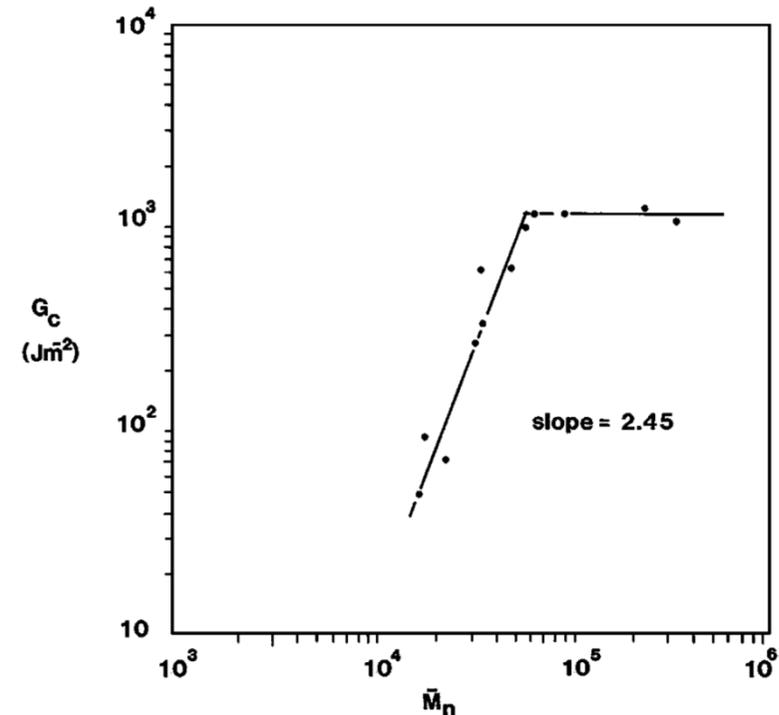
- GICs exhibit thermoplastic polymer behaviour
- Reptation pull-out model by Prentice⁸
- Polymer chain trapped in a tube of entanglements formed by neighbouring chains
- $G \propto (M_n)^2$ this implies the slope should be 2



Fracture of GLCs

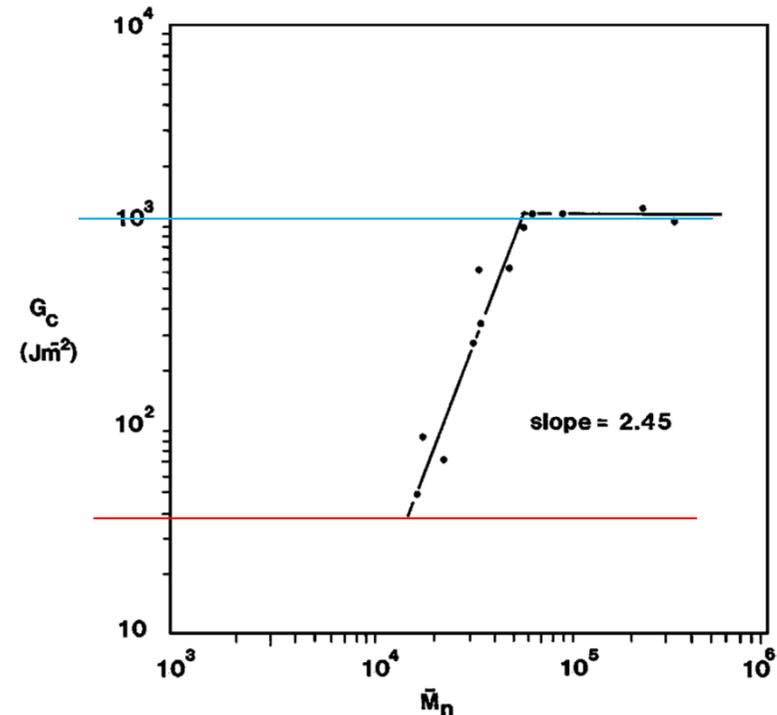
Assumptions

- Polymer crosses the fracture plane only once
- No distortion of the tube, distortion requires more work to remove the chain, increasing the plastic zone size
- Assumed that the polymer is monodisperse, whereas it has a polydisperse distribution



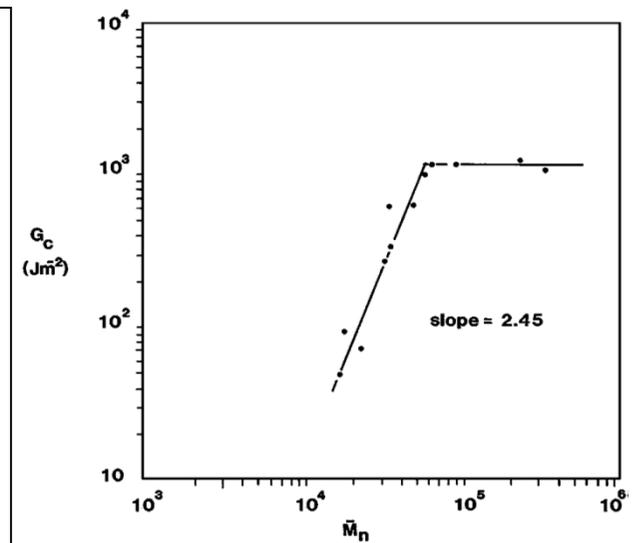
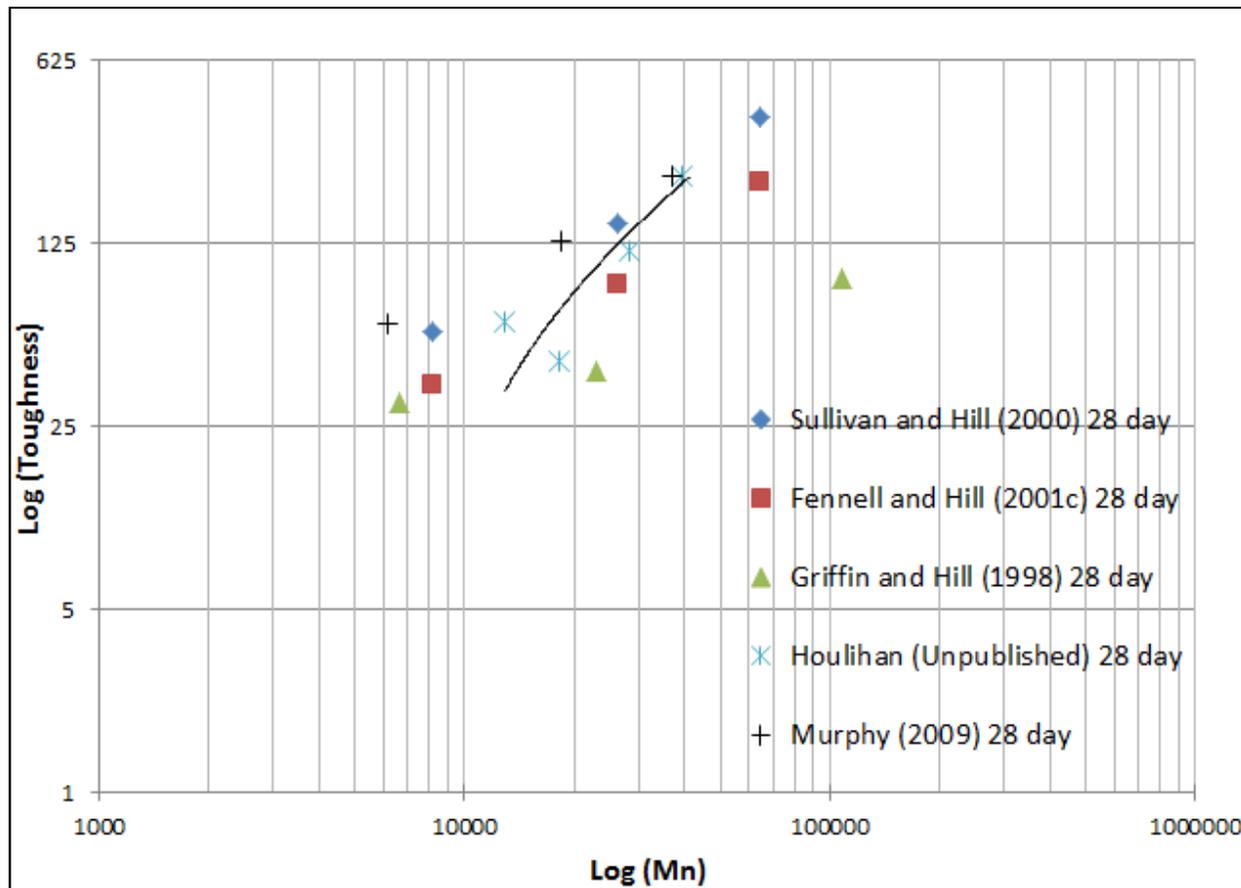
Fracture of GLCs

- Dependency of toughness on the M_n of the polymer
- Once the critical molar mass is reached, the toughness is independent of M_n
- Force to remove the chain from its tube is greater than that to break the C-C bond of the polymer backbone



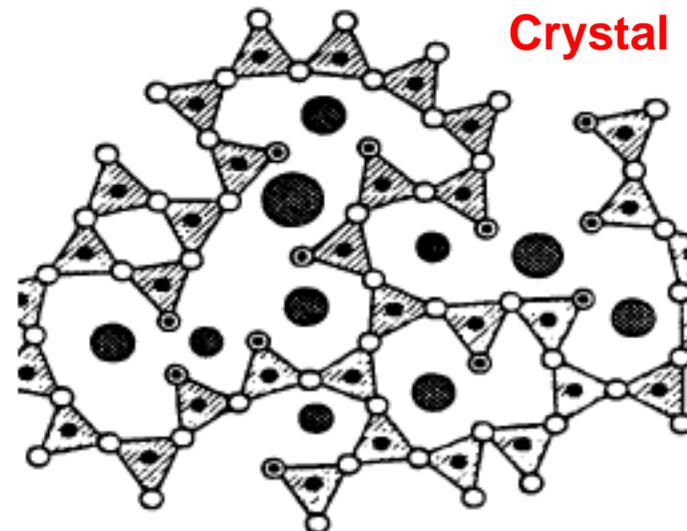
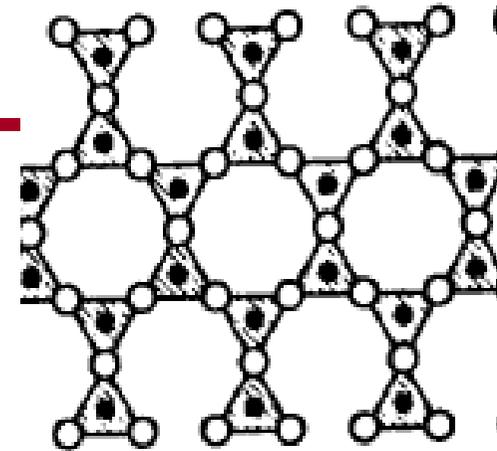
Effect of M_n

Toughness (G)



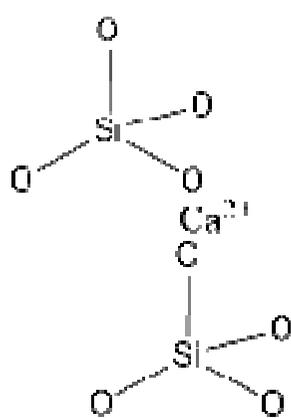
Glass Structure

- Crystal - Fixed Bond Angles and Distances. Regular Periodic Structure. Low Energy State.
- Glass – Variation in Bond Angles and Distances. Disordered and High Energy State. Exhibits a Glass Transition.
- Produced by rapid quenching of a molten liquid.

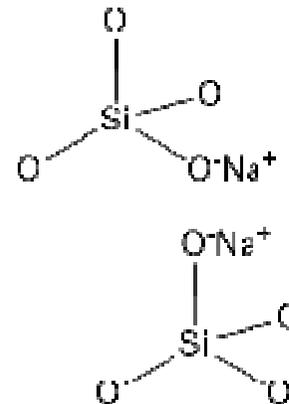


Example Ionomer Glass -Sodium Series (1993!)

$1.5\text{SiO}_2 \cdot 1.0\text{Al}_2\text{O}_3 \cdot 0.5\text{P}_2\text{O}_5 \cdot 1-X\text{CaO} \cdot 0.5\text{CaF}_2 \cdot X\text{Na}_2\text{O}$
where $X = 0.1, 0.2, 0.3$ and 0.4 .



(a)



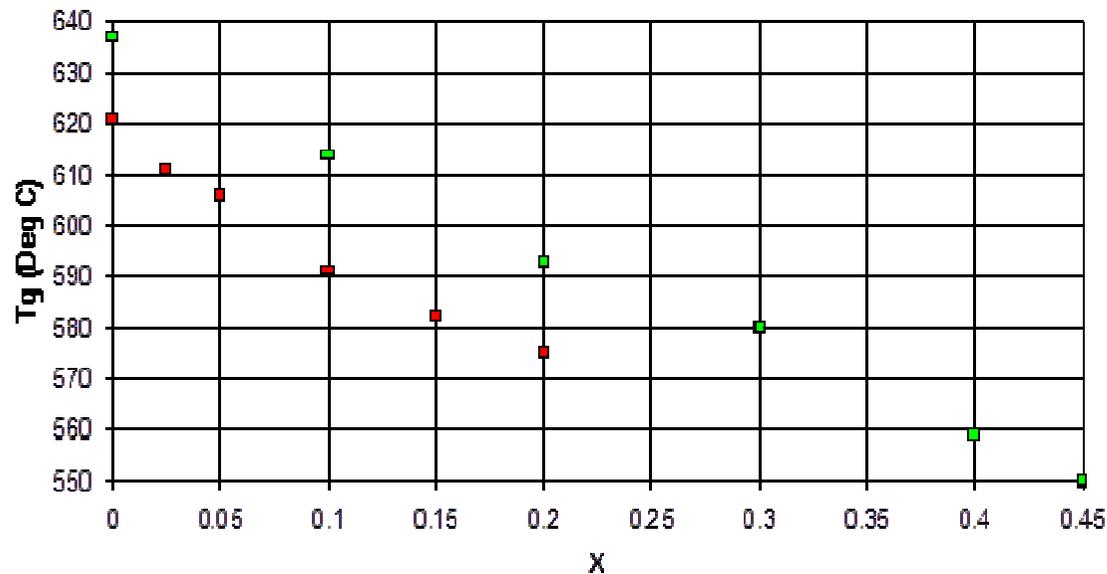
(b)

Hypothesis: Replacement of structure on lhs by that on rhs would reduce T_g and increase reactivity. Many aspects of structure unknown at the time (role of F) – Properties difficult to interpret at the time

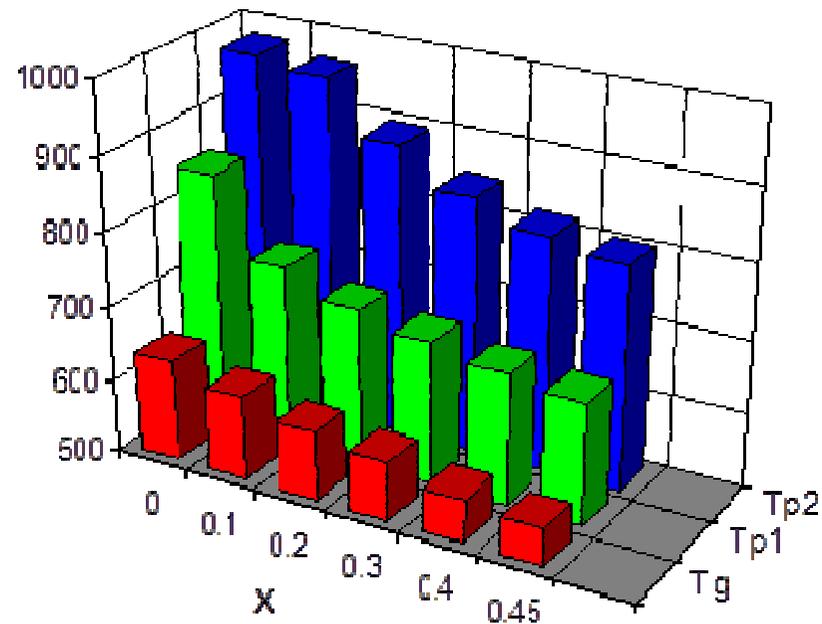
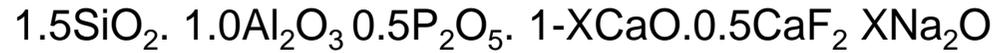
Example Ionomer Glass -Sodium Series



where $X = 0.1, 0.2, 0.3$ and 0.4 . Na = Green, K = Red



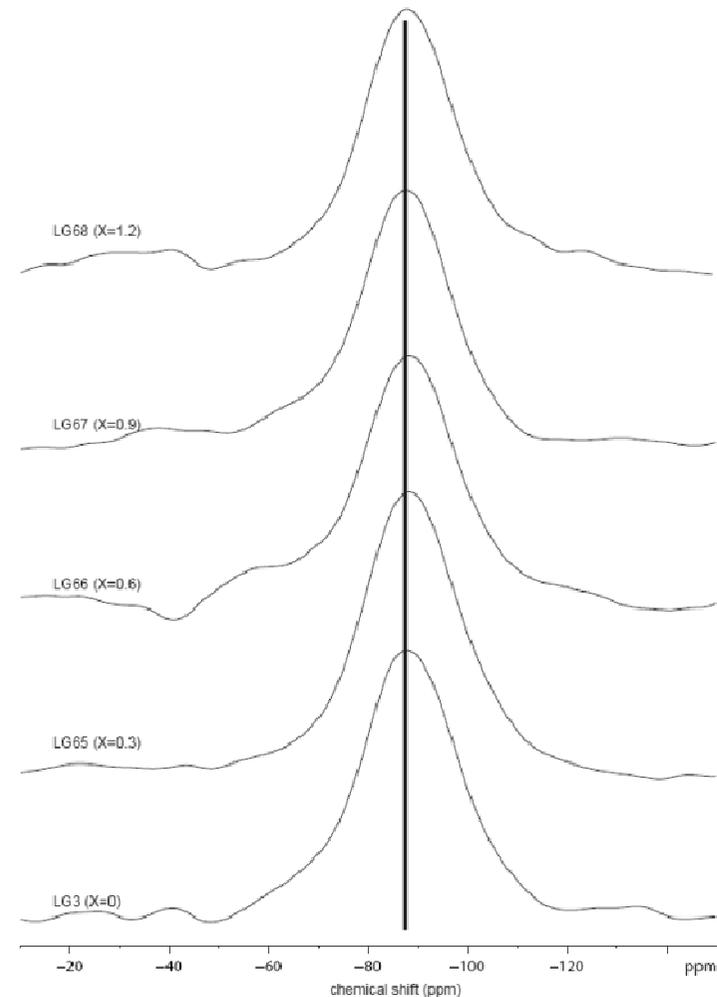
Example Ionomer Glass -Sodium Series DSC



Glass Transition Temperatures and First and Second Peak Crystallisation Temperatures For $1.5\text{SiO}_2 \cdot 0.5\text{P}_2\text{O}_5 \cdot \text{Al}_2\text{O}_3 \cdot X\text{R}_2\text{O} \cdot (1-X)\text{CaO} \cdot 0.5\text{CaF}_2$ Glasses

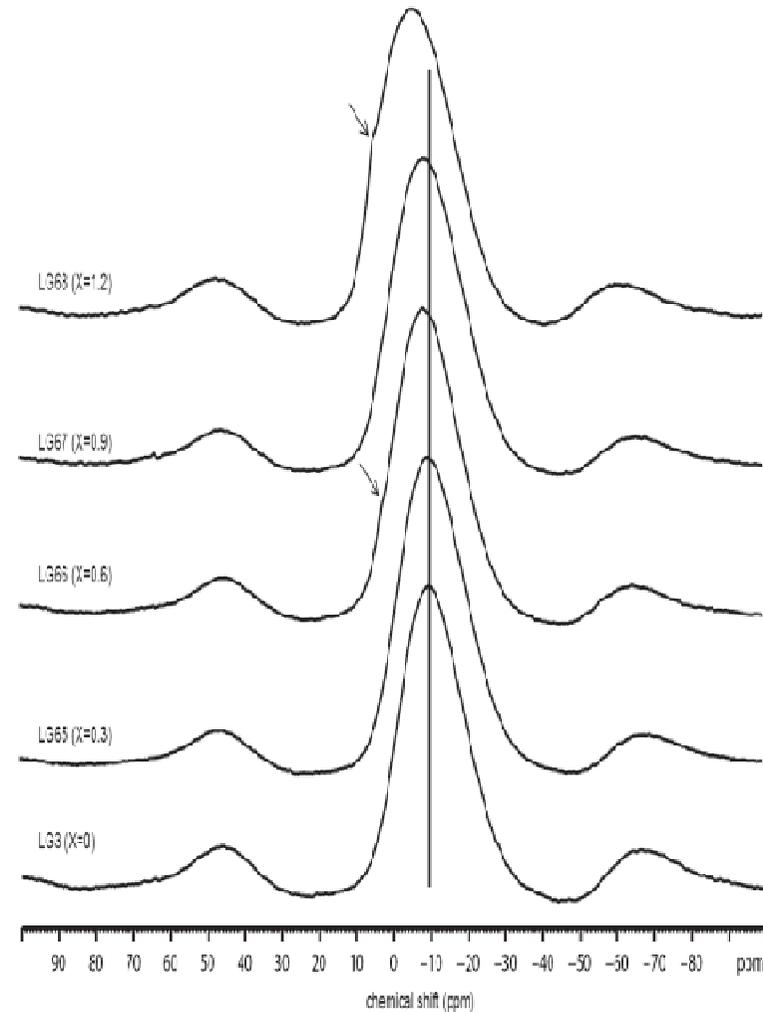
Example Ionomer Glass -Sodium Series ^{29}Si MAS-NMR

^{29}Si MAS-NMR spectra of sodium glasses
All glasses demonstrate the same
chemical shift at around -87.0 to -88.0
ppm. Predominantly Q^3 structure, but
unchanged by Na substitution



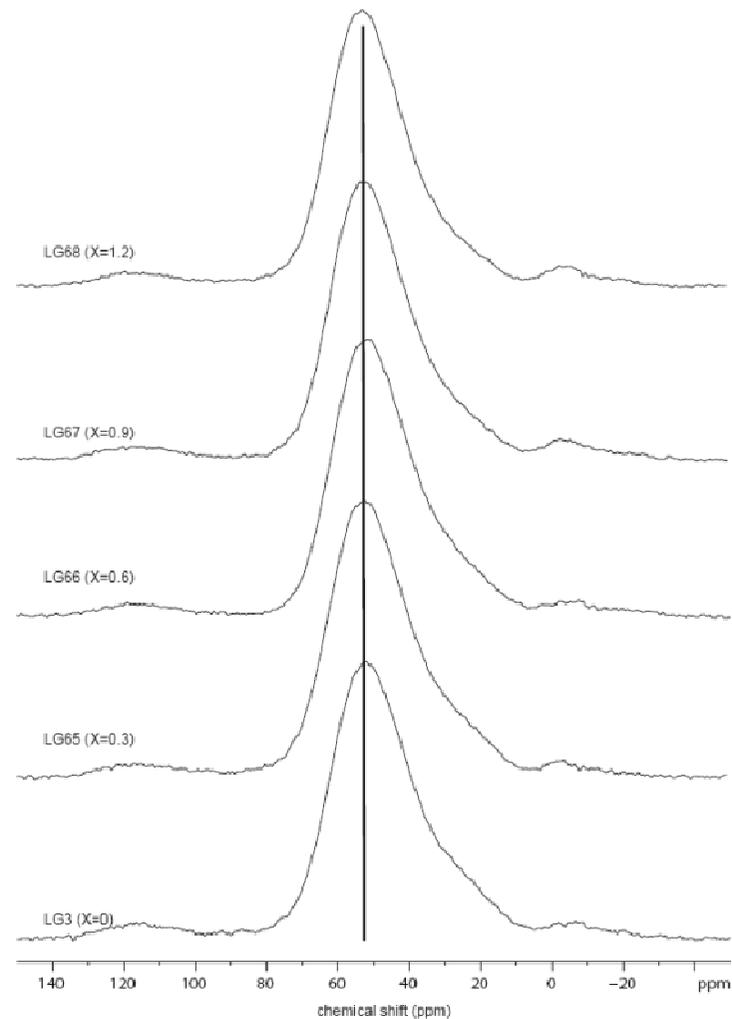
Example Ionomer Glass -Sodium Series ^{31}P MAS-NMR

^{31}P MAS-NMR spectra of sodium glasses of LG3, LG65, LG66, LG67 and LG68. Shows P is in Q^1 pyrophosphate role and with sodium increase shoulder appears (arrow) indicating some Q^0 orthophosphate formation

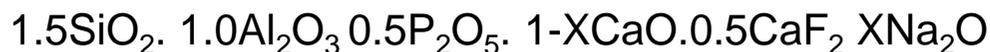


Example Ionomer Glass -Sodium Series ^{27}Al MAS-NMR

^{27}Al MAS-NMR spectra of sodium glasses
The chemical shift remains the same for all the glasses. The major peak at around 52.0-53.0 ppm is Al(IV) and there is also a shoulder for Al(V) sites and a small peak at -2.0 ppm for Al(VI) sites. The line is a guide to the eye only

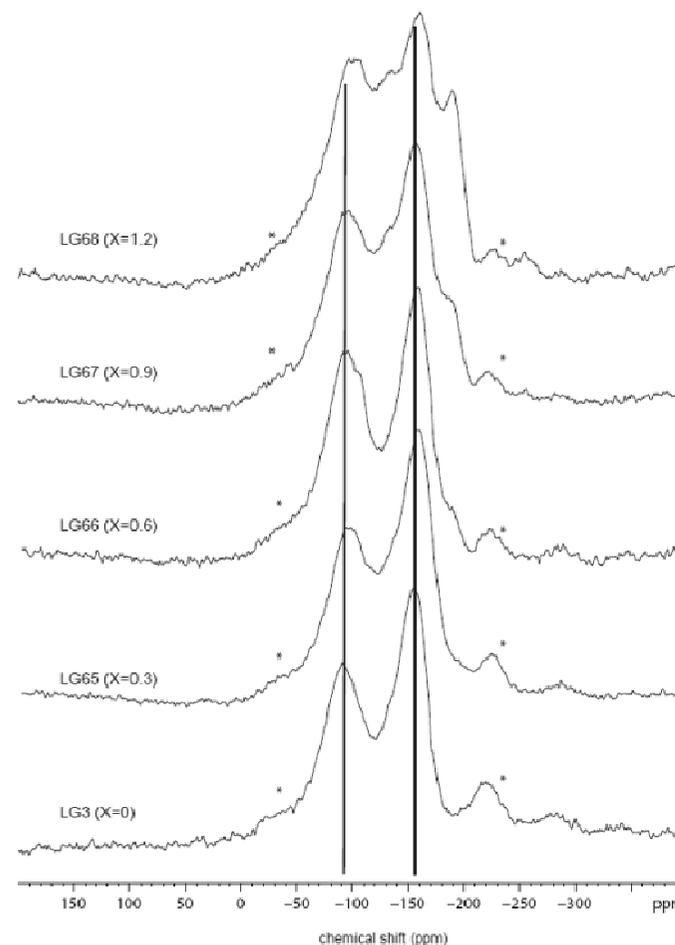


Example Ionomer Glass -Sodium Series



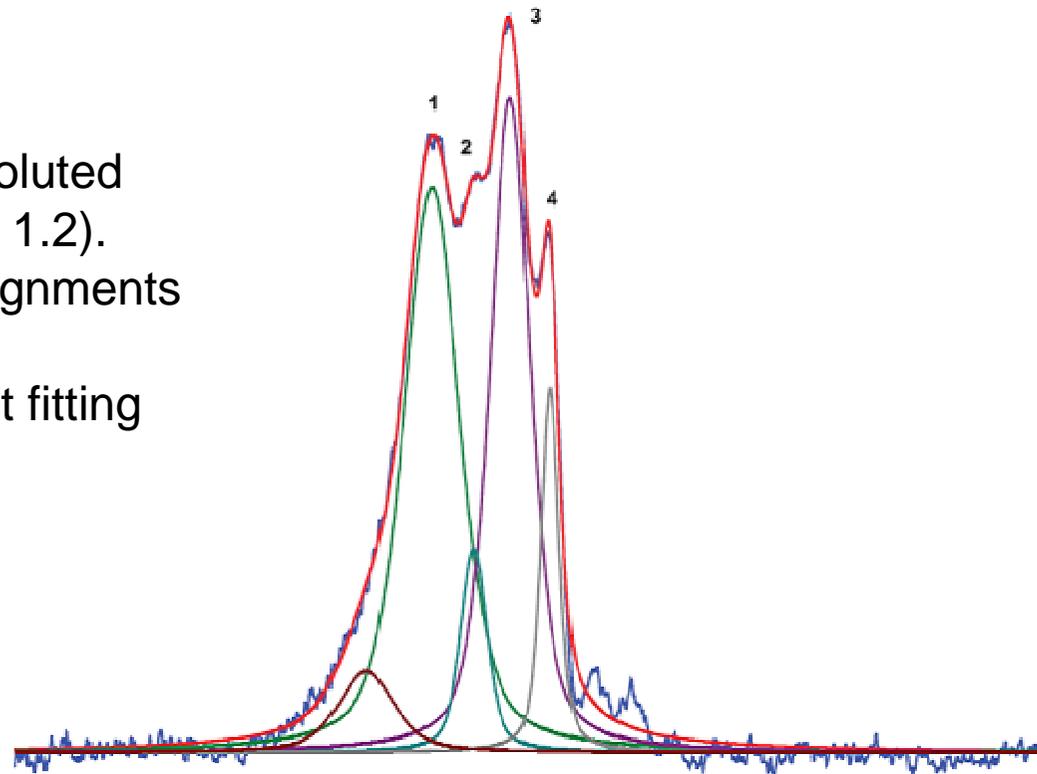
19F MAS-NMR spectra of sodium glasses of LG3, LG65, LG66, LG67 and LG68

There are two major peaks at -100.0 and -150.0 ppm corresponding to F-Ca(n) and Al-F-Ca(n). As the amount of sodium in the glass is increased, another two peaks appear at around -132.0 ppm which correspond to a mixed site of F-Ca/Na and at -186.0 ppm for Al-F-Na(n) site. The spinning sidebands are indicated by (*) and the lines are a guide to the eye only



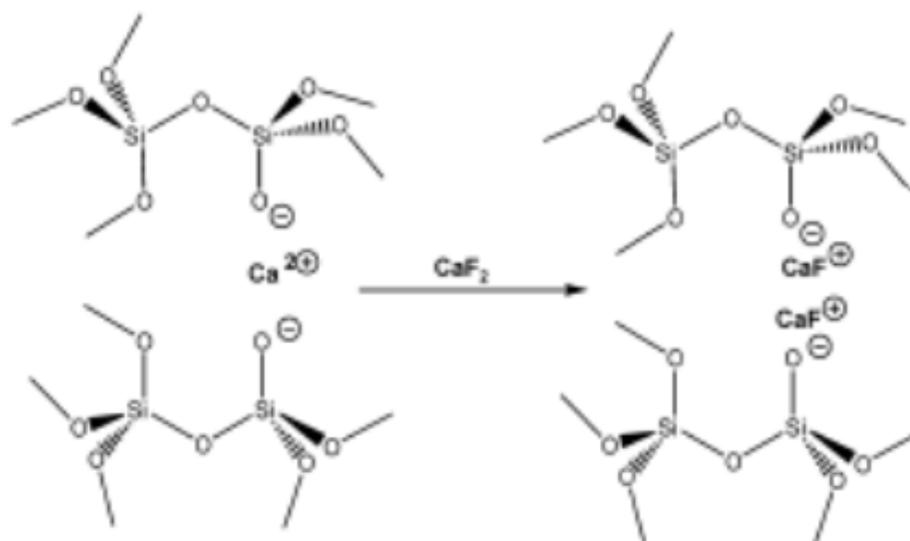
Example Ionomer Glass -Sodium Series

The experimental and the de-convoluted spectra of the LG68 glass ($\text{Na}_2\text{O} = 1.2$).
Peak labels correspond to the assignments given in Table
Fittings were performed using dmfit fitting program Gaussian model



Example Ionomer Glass -Sodium Series

Peak	Species	Chemical shift, ppm	Percentage
1	F-Ca(n)	-97.7	44.9
2	F-Ca(3)Na(1)	-130.2	7.6
3	Al-F-Ca(n)	-158.3	37.8
4	Al-F-Na(n)	-189.0	9.6



Example Ionomer Glass -Sodium Series

Conclusion:

Presence of fluorine in these glasses can be present as

- fluorine calcium complexes
- mixed fluorine calcium complexes
- non bridging fluorines as aluminium fluorine complexes charge balanced by either calcium or sodium.

Substitution of Ca by Na changes the ratio and type of these speciations

Structural roles in Ionomer and Bioactive Glasses

SiO₂

- increases network connectivity
- reduces bioactivity
- rate of network dissolution decreases

P₂O₅

- increases surface reactivity
- increases bioactivity
- increases degradation rate
- high conc's result in adverse effects

CaO, Na₂O

- reduce network connectivity

Structural roles in Ionomer and Bioactive Glasses

F

- increase bone formation at low conc's
- toxic at high conc's

Mg, K, B

- little effect on bioactivity

Al

- increases network connectivity
- can inhibit bone bonding
- increases resistance to ion exchange surface reactions
- interferes with osteoblast and fibroblast metabolism

Ta, Ti, Sb, Zr

- increase network connectivity

Predicting the bioactivity of glasses using the network connectivity or split network models

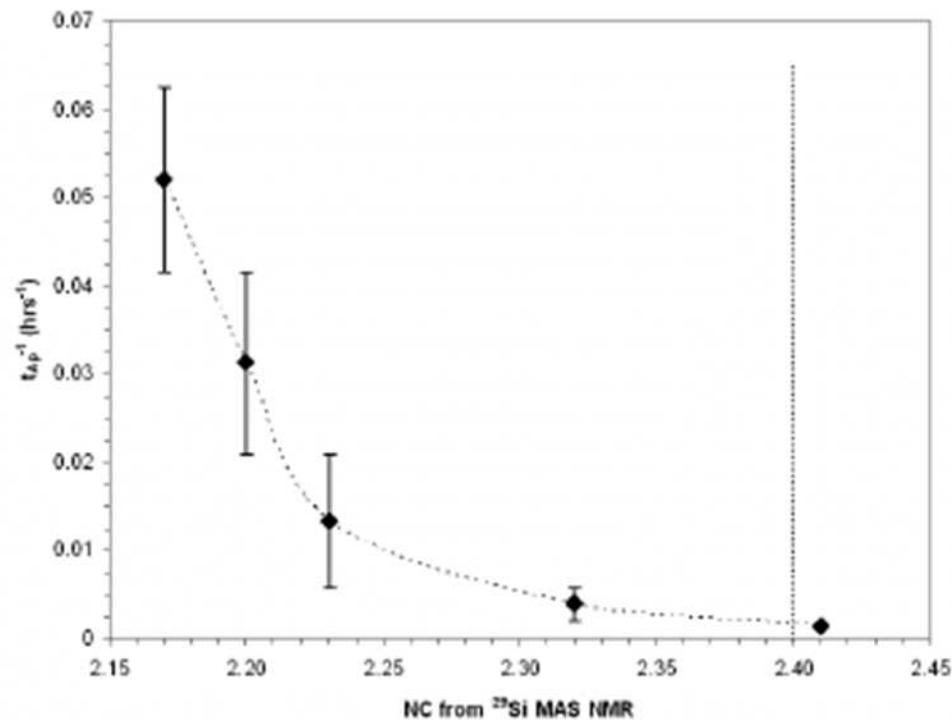
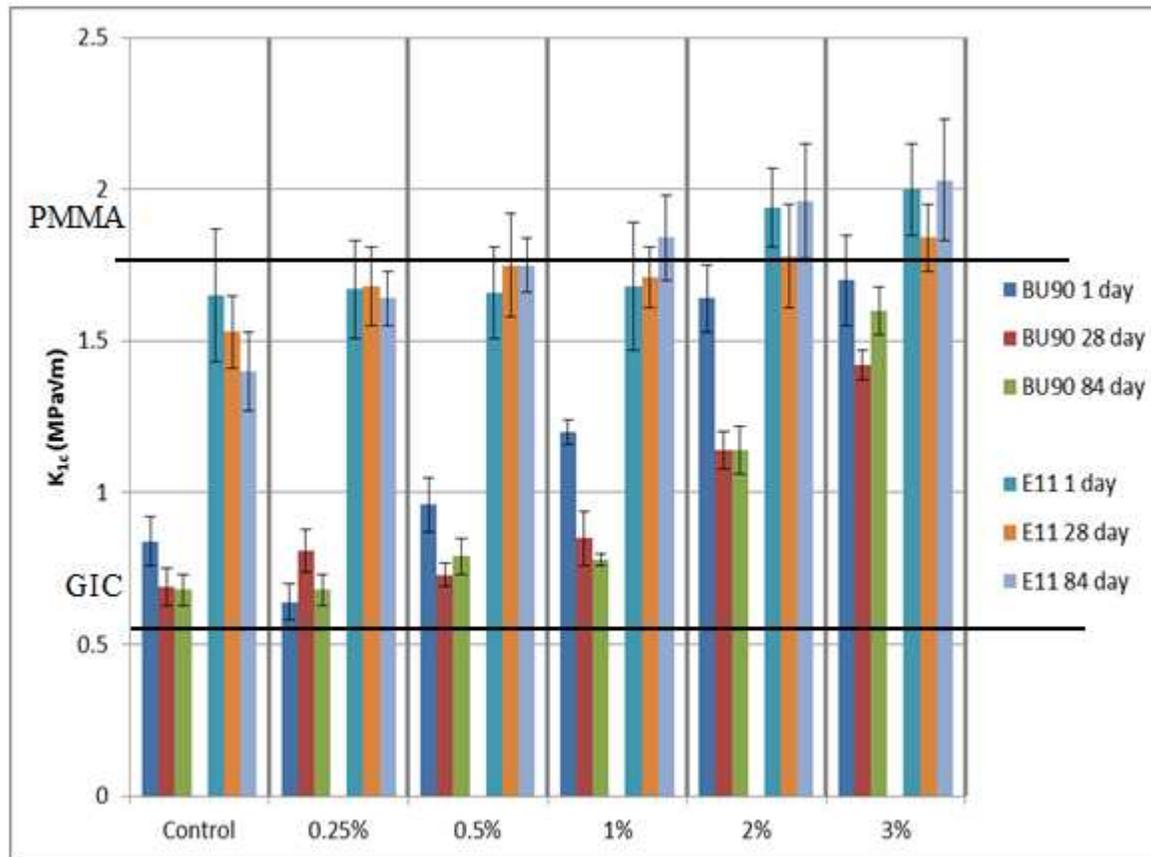


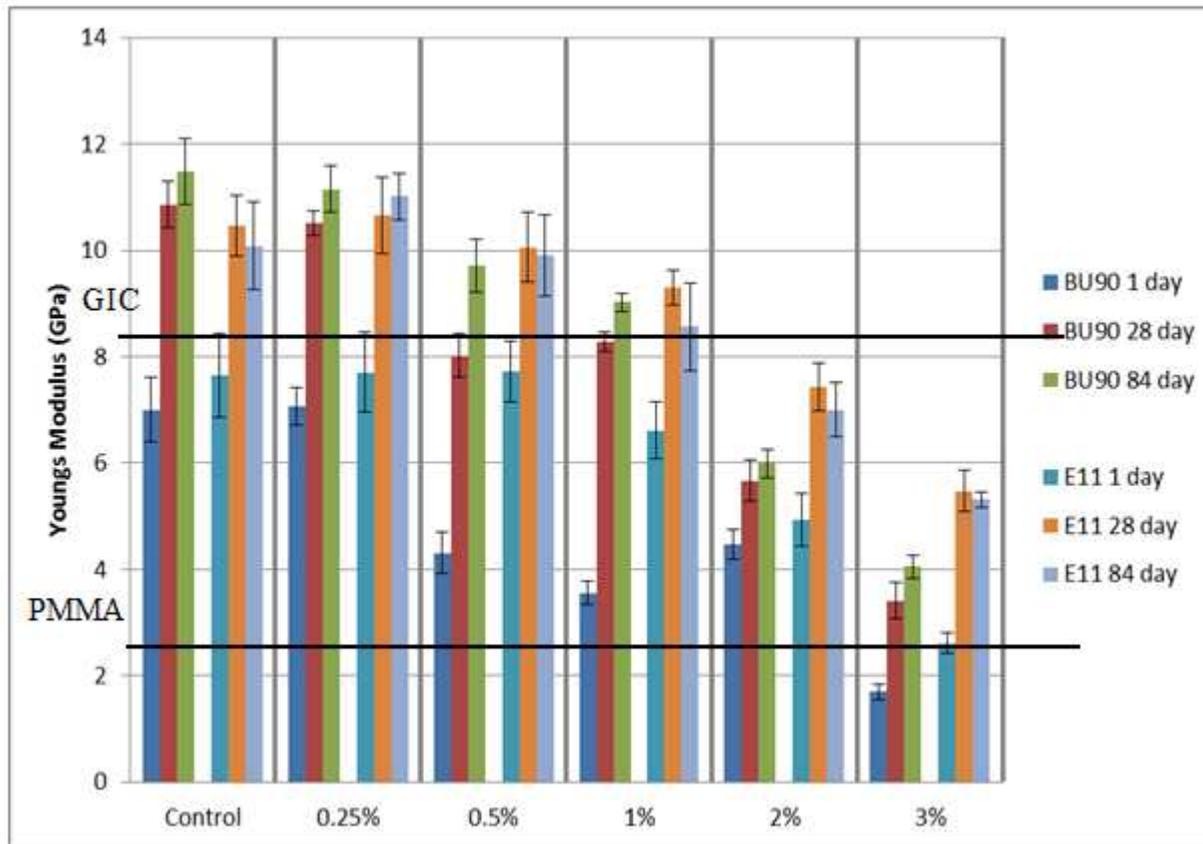
Figure 1: Bioactivity (defined as t_{Ap}^{-1} , where t_{Ap} is the time of first apatite formation in SBF as detected by XRD) of BG in the Mg series vs. NC_{NMR} calculated from the proportions of Q^2 and Q^3 silicons from ^{29}Si MAS NMR [11]. The vertical line represents the percolation point ($NC = 2.4$), i.e. the cut-off value for bioactivity as defined by Hill [7].

Results



- K_{1c} of commercial GIC exceeded
- Matched K_{1c} of PMMA in some cases

Results



- Maintained E compared to GIC
- Matched E across formulations compared to PMMA

Mechanical data

	Cortical Bone	GIC from present study	Commercial GIC Fuji IX	PMMA
Fracture toughness (MPa√m)	6.4 ¹	0.52 – 2.3	0.35 - 0.56 ²	1.82 - 3.52 ⁵
Young's Modulus (GPa)	15 – 30 ³	2.27 – 15.61	7 - 8 ³	1.90 - 2.28 ⁵
Toughness (J/m ²)	3400 ¹	44.58 – 1281.75	40 ^{2,3}	1521.58 – 5437.27 ⁵
Flexural Strength (MPa)	170 ¹	14.1 – 53.5	17.8 – 51 ^{2,4}	49.94 – 69.46 ⁵

¹Ziopoulos and Currey (1998) Changes in the stiffness, strength, and toughness of human cortical bone with age. *Bone*, 22, 57-66.

²Moshaverinia et al., (2010) Measure of microhardness, fracture toughness and flexural strength of N-vinylcaprolactam (NVC)-containing glass-ionomer dental cements. *Dental materials*, 26, 1137-1143.

³Ratner et al., (2004) *Biomaterials Science: An Introduction to Materials in Medicine*, Elsevier Science.

⁴Lohbauer et al., (2010) Dental Glass Ionomer Cements as Permanent Filling Materials – Properties, Limitations and Future Trends. *Materials*, 3, 76-96

⁵Chilikina (unpublished) personal communication

Example Ionomer Glass –Fluoride Series

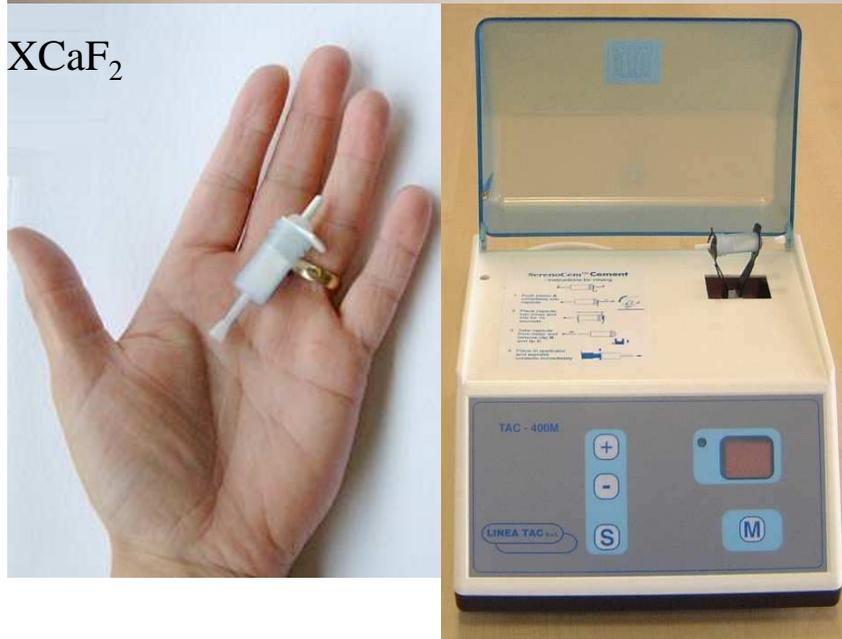


<u>GLASS CODE</u>	X
LG45	0.00
LG44	0.25
LG3	0.50
LG26	0.66
LG2	0.75
LG42	1.00

Serenocem™ Capsules



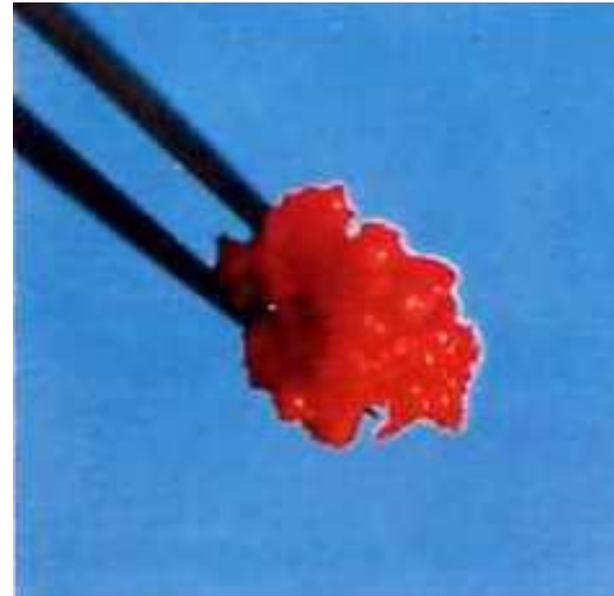
$1.5\text{SiO}_2 \text{ Al}_2\text{O}_3 \text{ } 0.5\text{P}_2\text{O}_5 \text{ CaO XCaF}_2$



Serenocem™ Granules



EM scan of granules, showing micro pores. magnification x 20

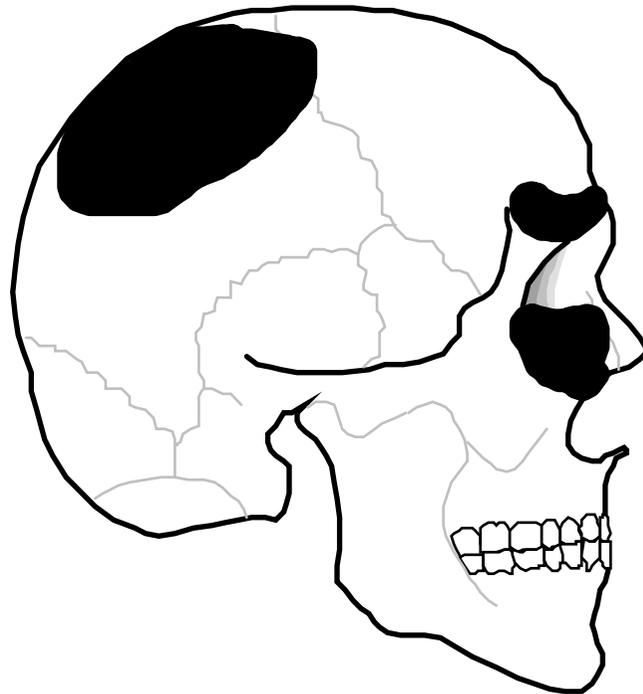


Hydrophilic granules absorb blood to produce fibrin clot.

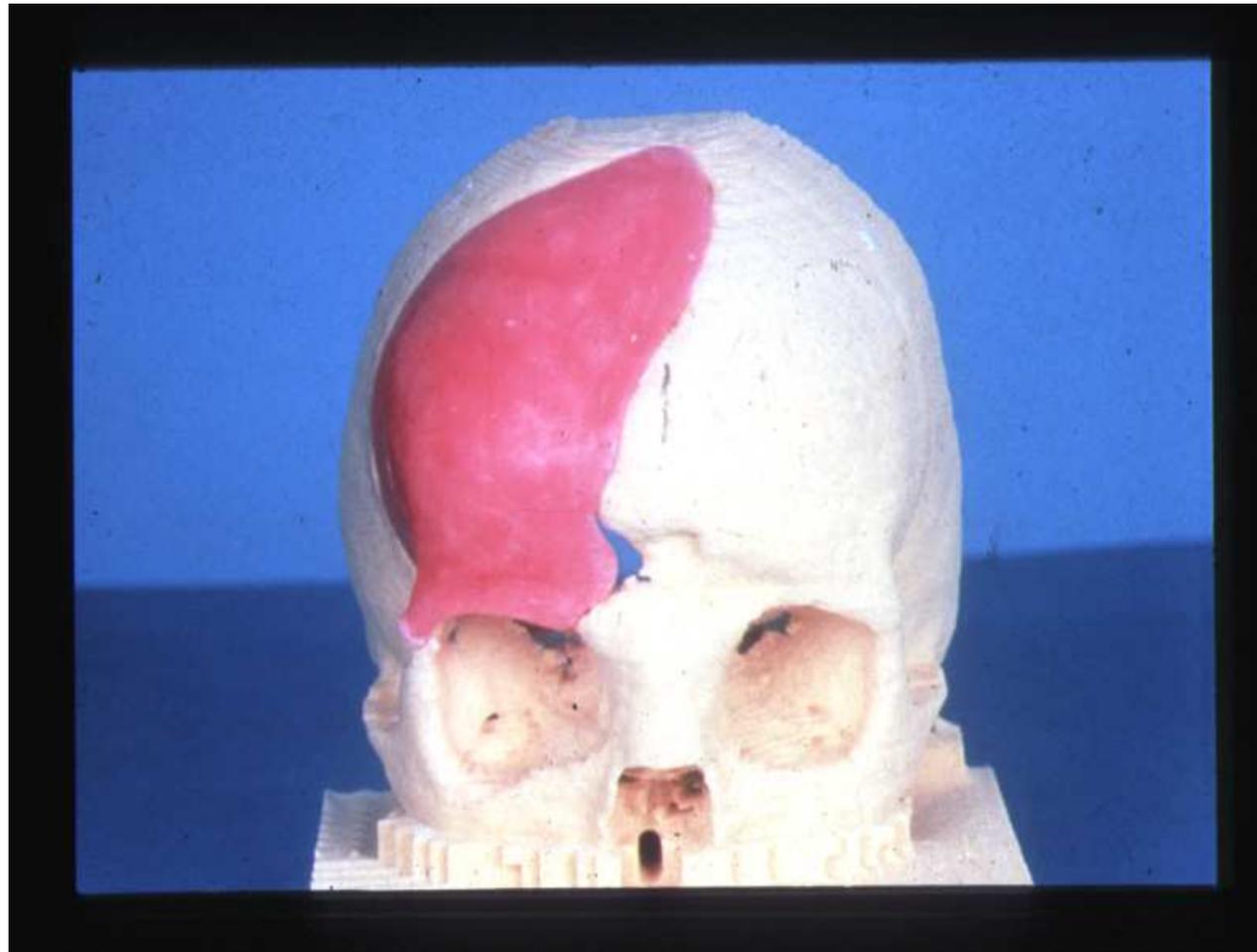
Produced by incorporating CaCO_3 into the cement which generates CO_2 in-situ, foaming the cement . Sold as a cancellous bone substitute.

GPA Cement BONE SUBSTITUTES

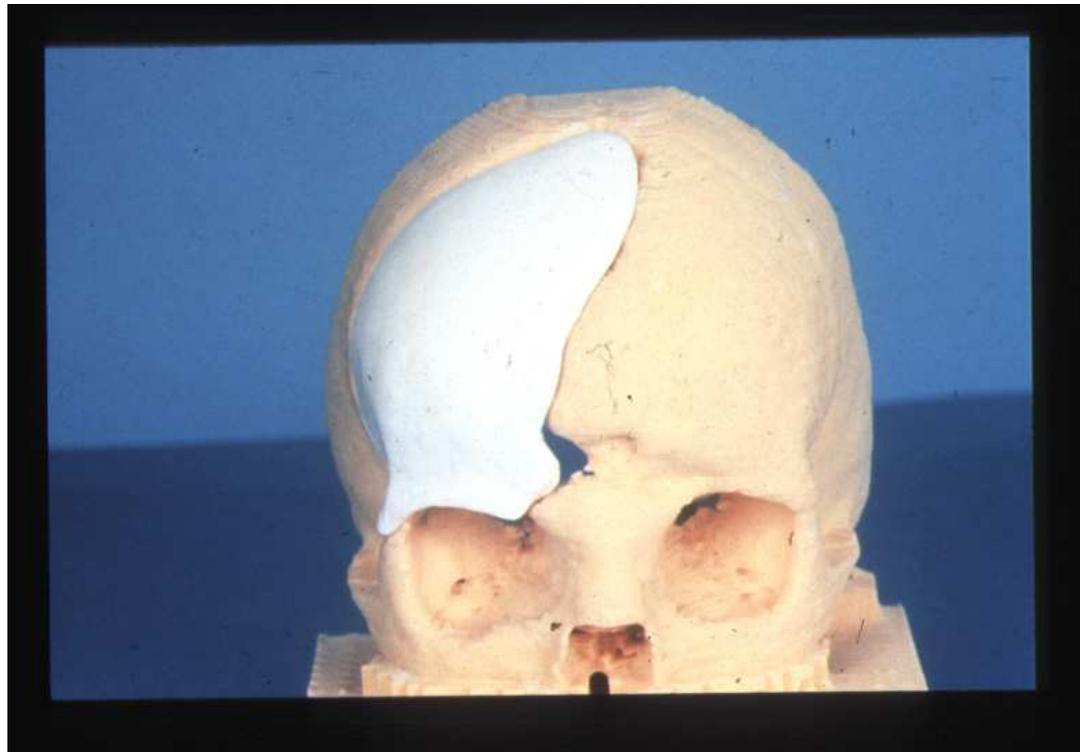
- e.g. (a) Cranial bone plates
- (b) Maxillofacial implants



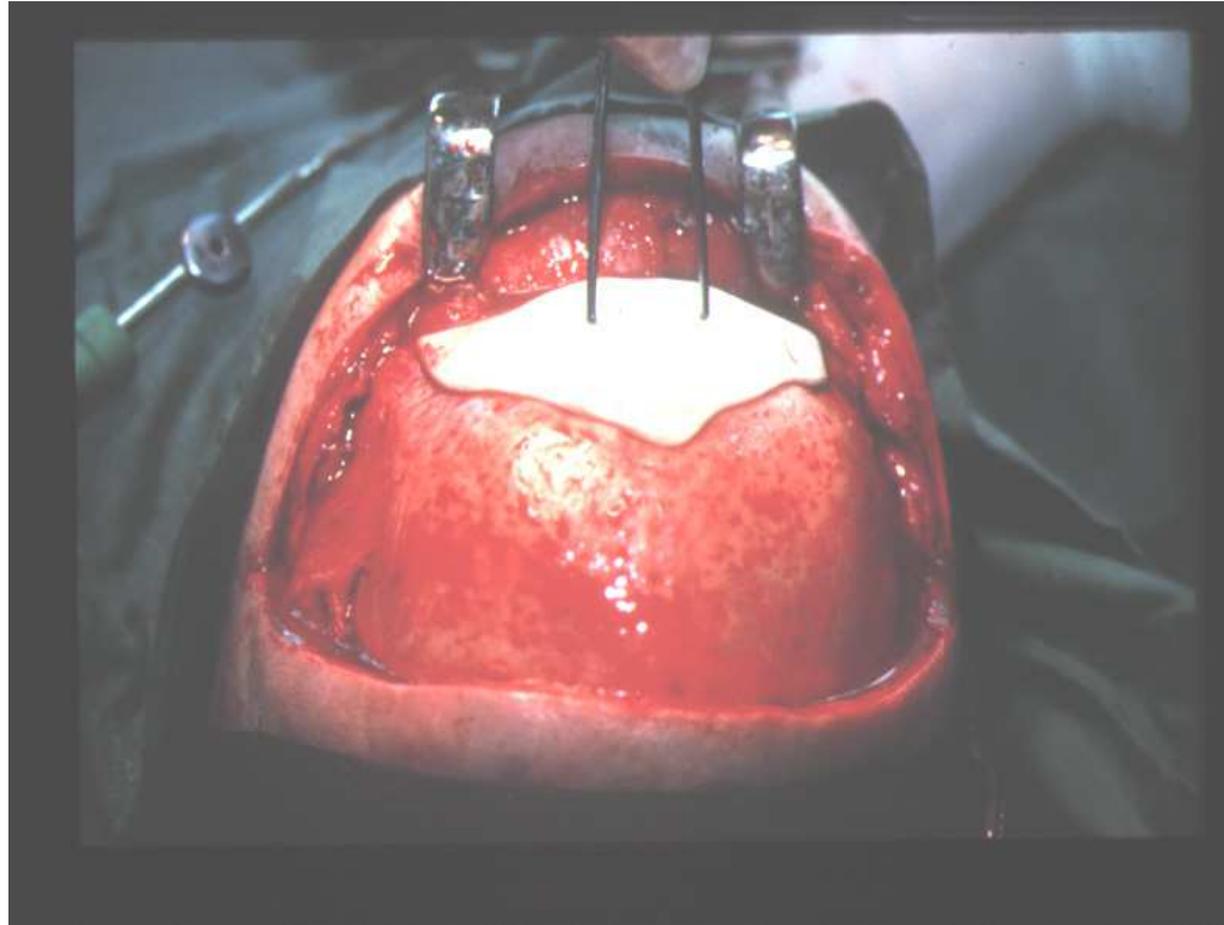
Wax Impression formed via CAT Scan



Custom Plate formed + Autoclaved



Placement of GI Cranial Plate



NB This product off market

Glass Ionomer cements

- Advantages
 - Non-exothermic setting reaction
 - Adhesive bond formed with bone and metals
 - Bioactive ions incorporated
 - Low systemic toxicity
 - Reduced local toxicity
- Disadvantages
 - Inferior mechanical properties
 - Neurotoxicity of Aluminium and PAA

